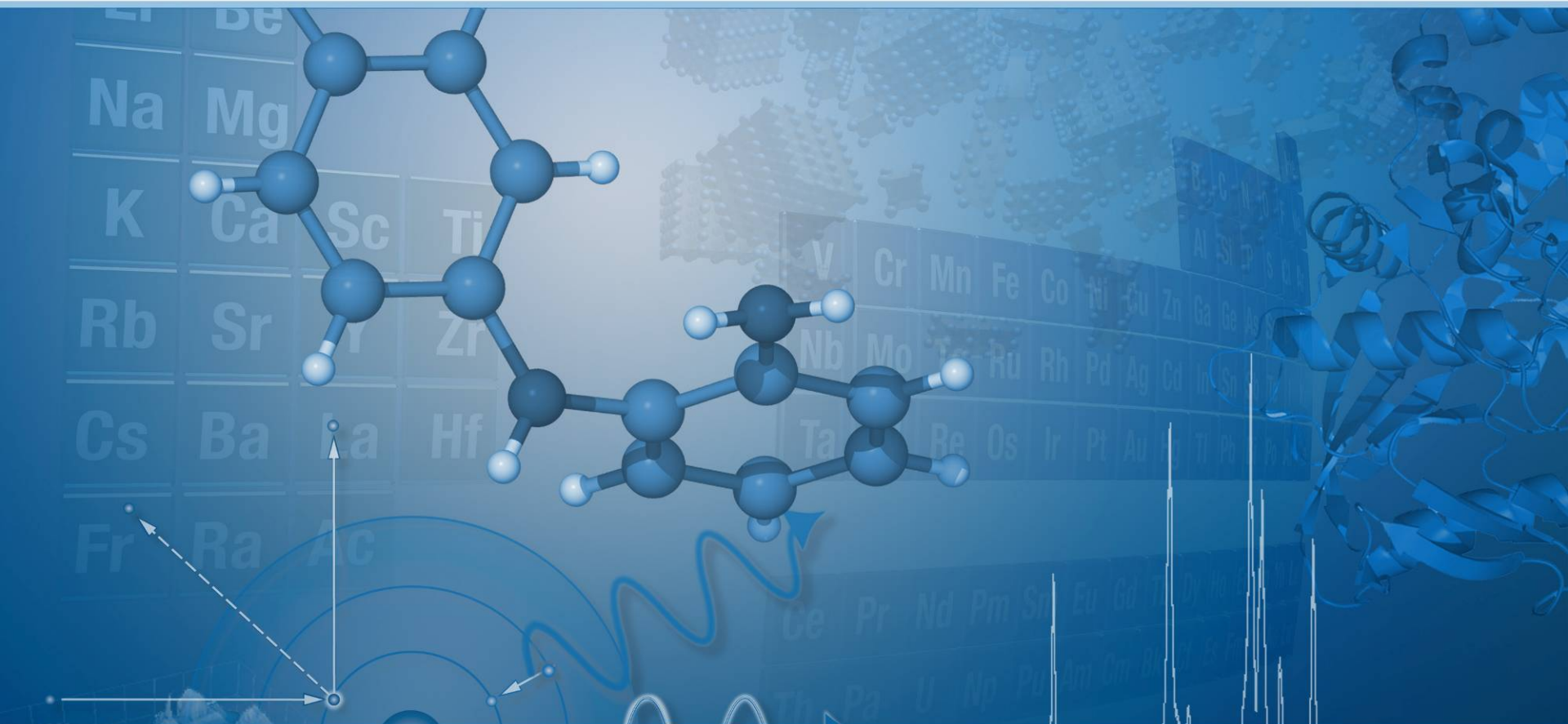


# Good Diffraction Practice Webinar Series



In-Plane Grazing Incidence Diffraction – March 23, 2013

[www.bruker-webinars.com](http://www.bruker-webinars.com)



# Welcome



**Dr. Heiko Ress**

Global Marketing Manager  
Bruker AXS Inc.  
Madison, Wisconsin, USA  
[heiko.ress@bruker-axs.com](mailto:heiko.ress@bruker-axs.com)  
+1.608.276.3000



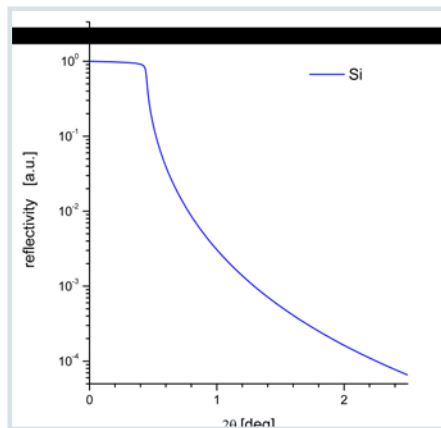
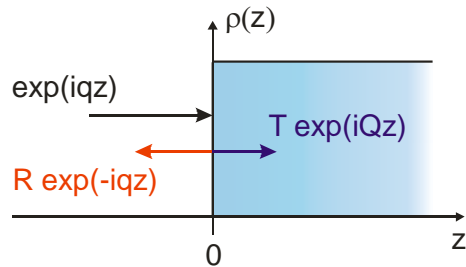
**Dr. Martin Zimmermann**

Applications Scientist, XRD  
Bruker AXS GmbH  
Karlsruhe, Germany  
[martin.zimmermann@bruker-axs.de](mailto:martin.zimmermann@bruker-axs.de)  
+49.721.50997.5602

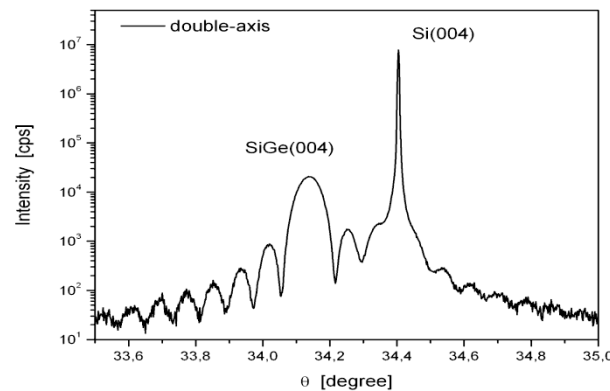
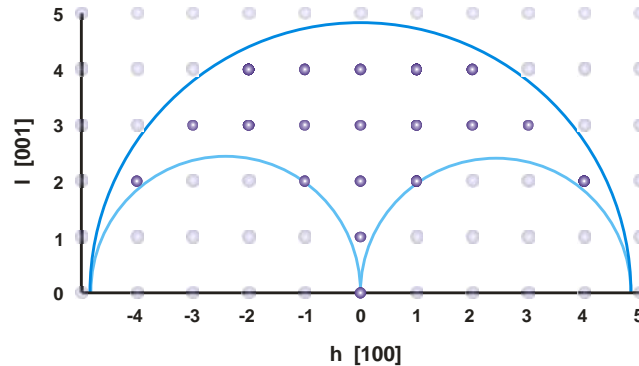
# Good Diffraction Practice Webinar Series History



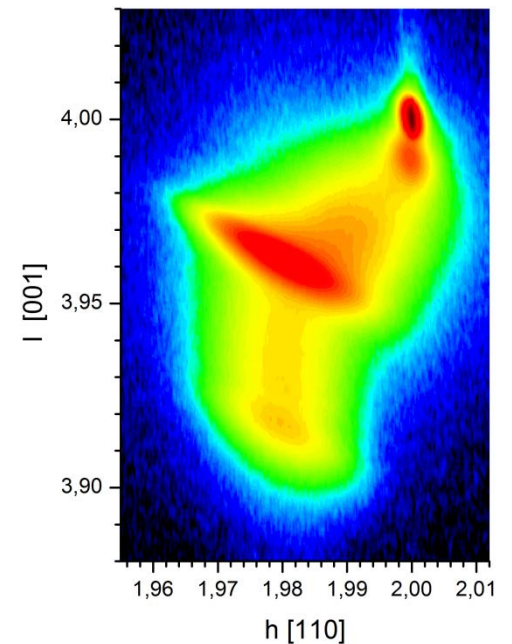
July 2010  
X-ray Reflectometry



May 2011  
High-Resolution X-ray  
Diffraction (HRXRD)



Jan 2012  
HRXRD – Reciprocal  
Space Mapping



# Outline

- Introduction
- Experimental configurations
- Experimental tips
- Examples

- Introduction
- Experimental configurations
- Experimental tips
- Examples

# What is In-Plane Grazing Incidence X-Ray Diffraction (IPGID)?



An X-ray scattering technique

- Non-destructive method
- X-rays probe on the nanometer scale

Diffraction technique

- Requires a crystal lattice
- Works for epitaxial and polycrystalline samples

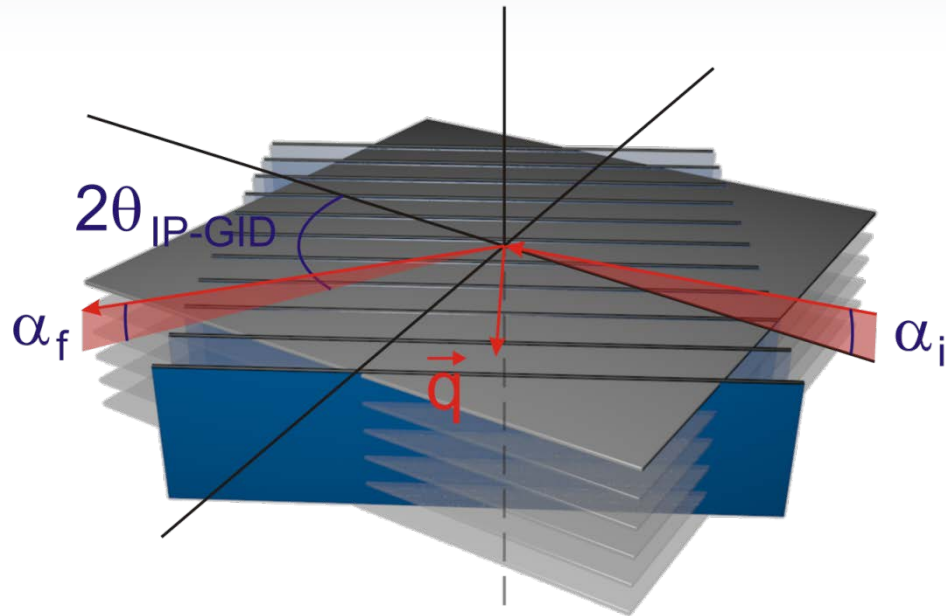
In-plane grazing incidence geometry

- Probes the near-surface part of the sample
- Probes the crystal properties parallel to the surface

What kind of **information** does IPGID provide about **my sample**?

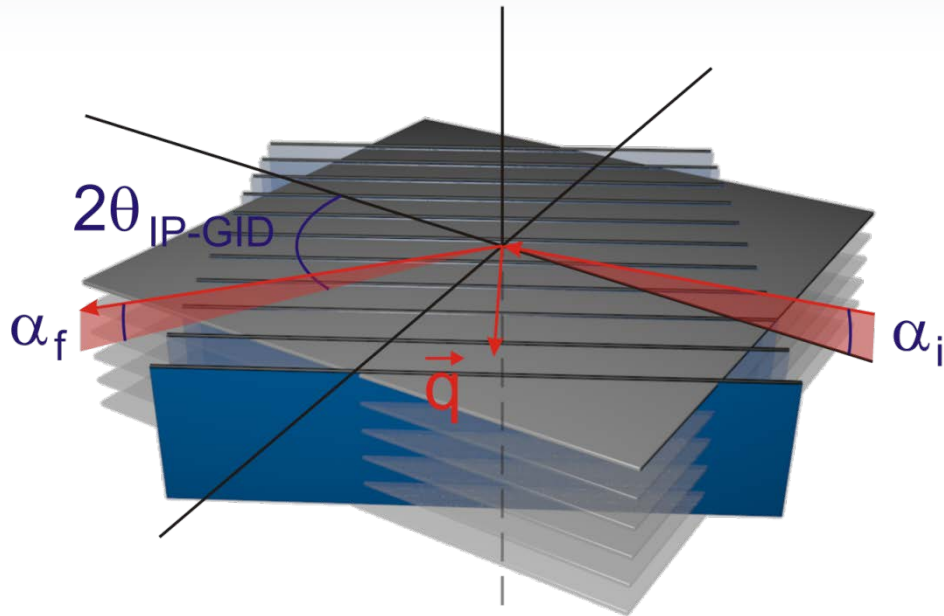
- In-plane lattice parameter
- Epitaxial relation
- Domain formation and twist
- In-plane texture
- Crystallite size
- Micro strain

# In-plane Grazing Incidence Diffraction: The scattering geometry



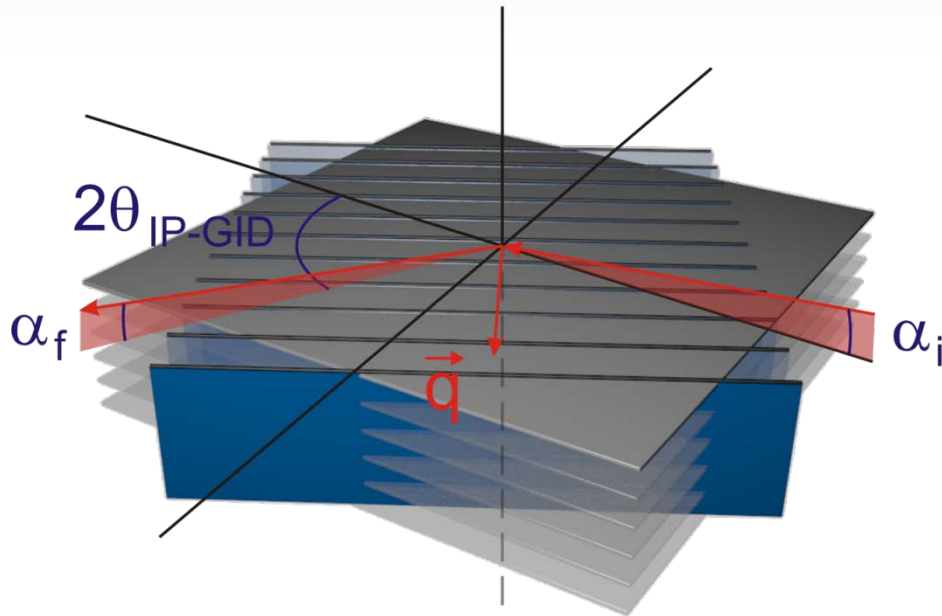


# In-plane Grazing Incidence Diffraction: The scattering geometry



Intensity :  $I(\vec{q}, \alpha_i, \alpha_f) \propto |T(\alpha_i)|^2 |F(\vec{q})|^2 |T(\alpha_f)|^2$

# In-plane Grazing Incidence Diffraction: The scattering geometry



Probed quantity :

$$|F(\vec{q})|^2 \propto \left| \int_V \rho(\vec{r}) \exp(i\vec{q}\vec{r}) d\vec{r} \right|^2$$

Transmission of  
incident and exit beam



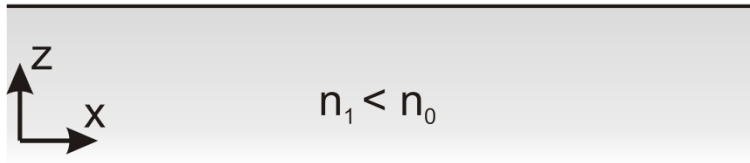
Intensity :

$$I(\vec{q}, \alpha_i, \alpha_f) \propto |T(\alpha_i)|^2 |F(\vec{q})|^2 |T(\alpha_f)|^2$$

# Reflectivity and Transmission of a substrate

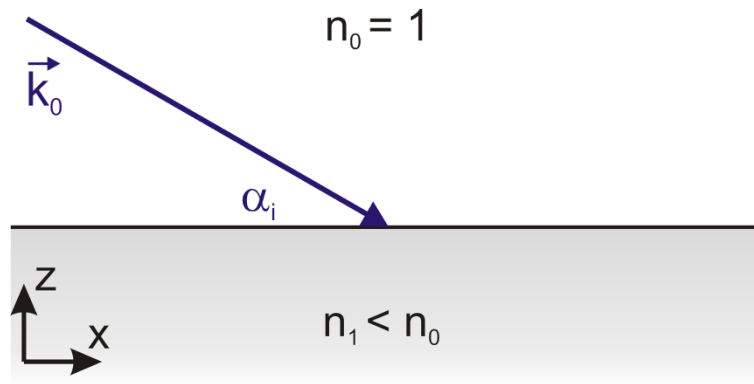


$$n_0 = 1$$



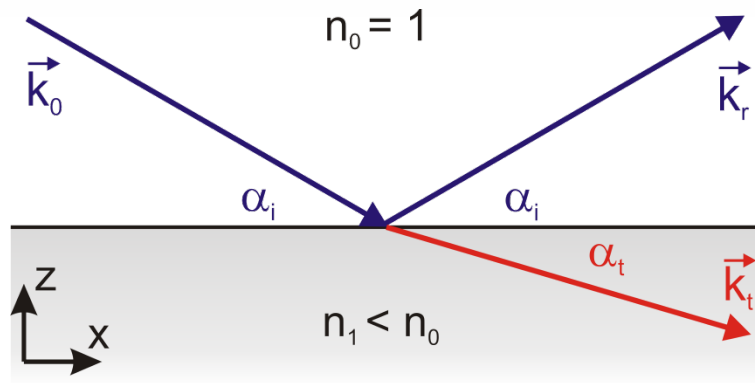
- Index of refraction:  $n = 1 - \delta + i\beta$
- For X-rays, dispersion:  $\delta \approx 10^{-5} - 10^{-6}$
- Absorption:  $\beta \approx (0.1, \dots, 0.01)\delta$

# Reflectivity and Transmission of a substrate



- Index of refraction:  $n = 1 - \delta + i\beta$
- For X-rays, dispersion:  $\delta \approx 10^{-5} - 10^{-6}$
- Absorption:  $\beta \approx (0.1, \dots, 0.01)\delta$

# Reflectivity and Transmission of a substrate



Reflection coefficient:

$$r = \frac{k_{0,z} - k_{t,z}}{k_{0,z} + k_{t,z}}$$

Transmission coefficient:

$$t = \frac{2k_{0,z}}{k_{0,z} + k_{t,z}}$$

- Index of refraction:  $n = 1 - \delta + i\beta$
- For X-rays, dispersion:  $\delta \approx 10^{-5} - 10^{-6}$
- Absorption:  $\beta \approx (0.1, \dots, 0.01)\delta$

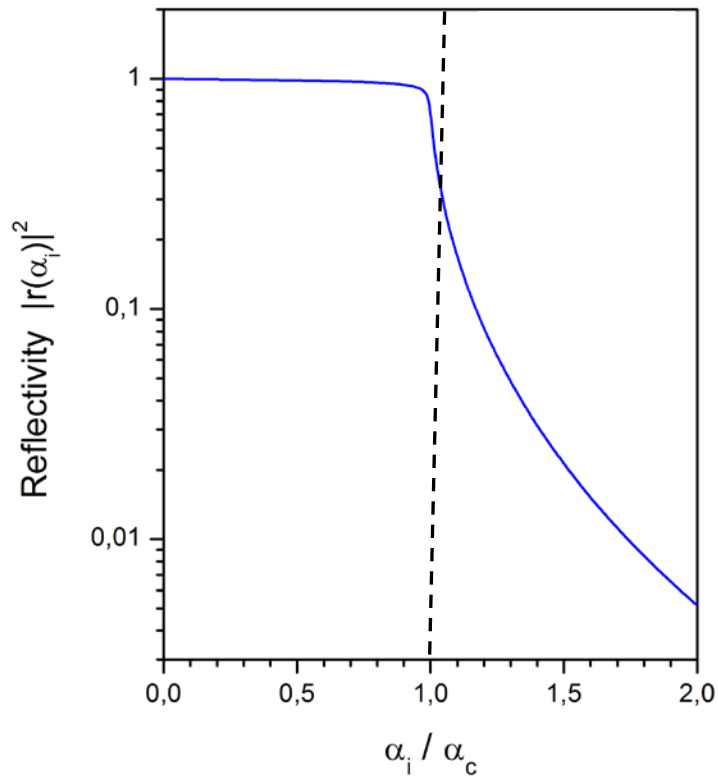
with  $k_{0,z} = k \sin \alpha_i$

$$k_{t,z} = k \sqrt{n^2 - \cos^2 \alpha_i}$$

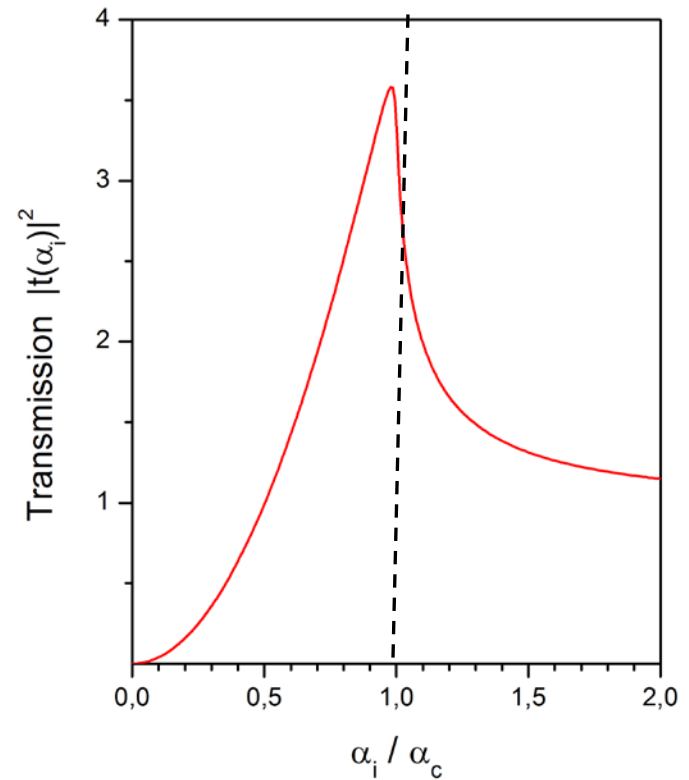
# Reflectivity and Transmission of a silicon substrate



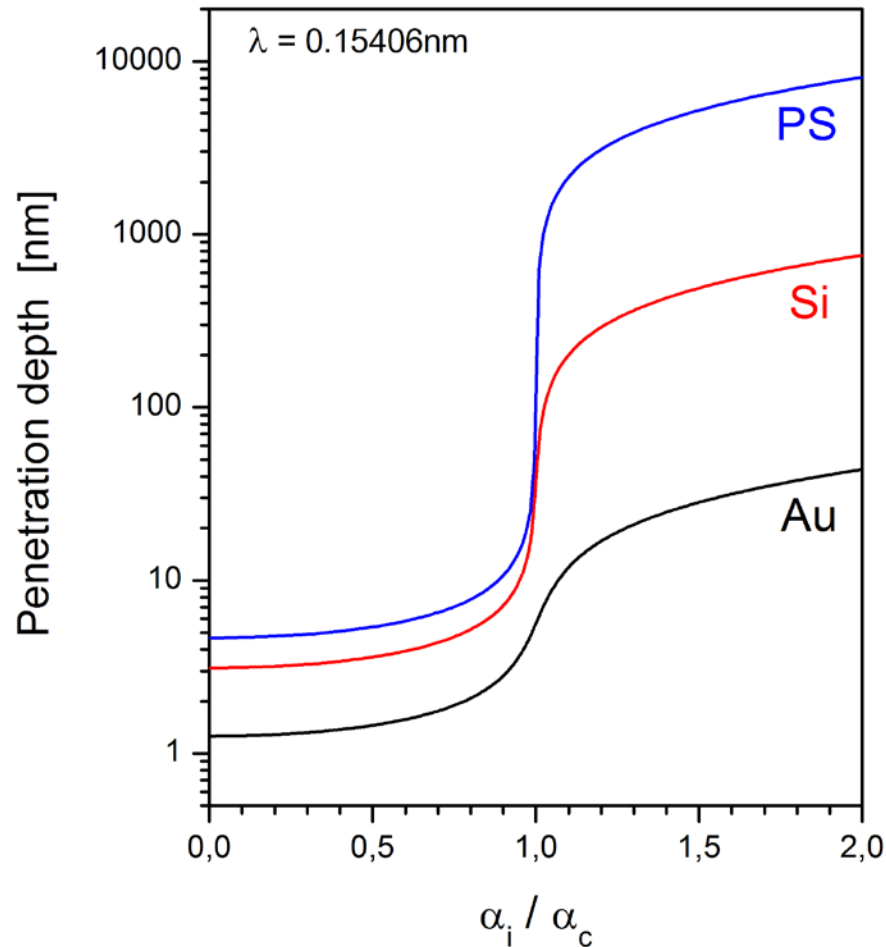
- Reflectivity



- Transmission



# Penetration depth for different materials



- Minimum penetration depths

$$\Lambda_0 = \frac{1}{\sqrt{4\pi r_e \rho}}$$

$r_e$  classical electron radius  
 $\rho$  electron density

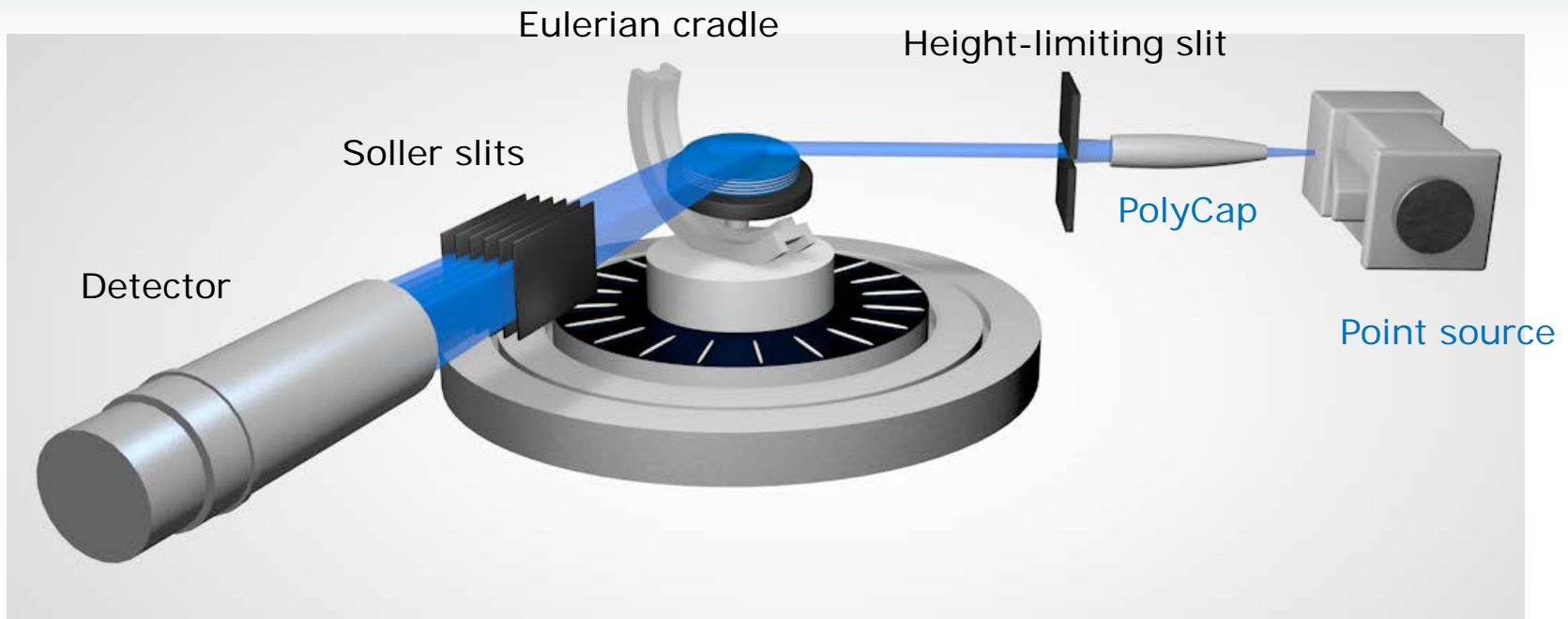
- Maximum penetration at high angles

$$\Lambda_{max} = \frac{\lambda}{2\pi\sqrt{\beta}}$$

- Introduction
- Experimental configurations
- Experimental tips
- Examples

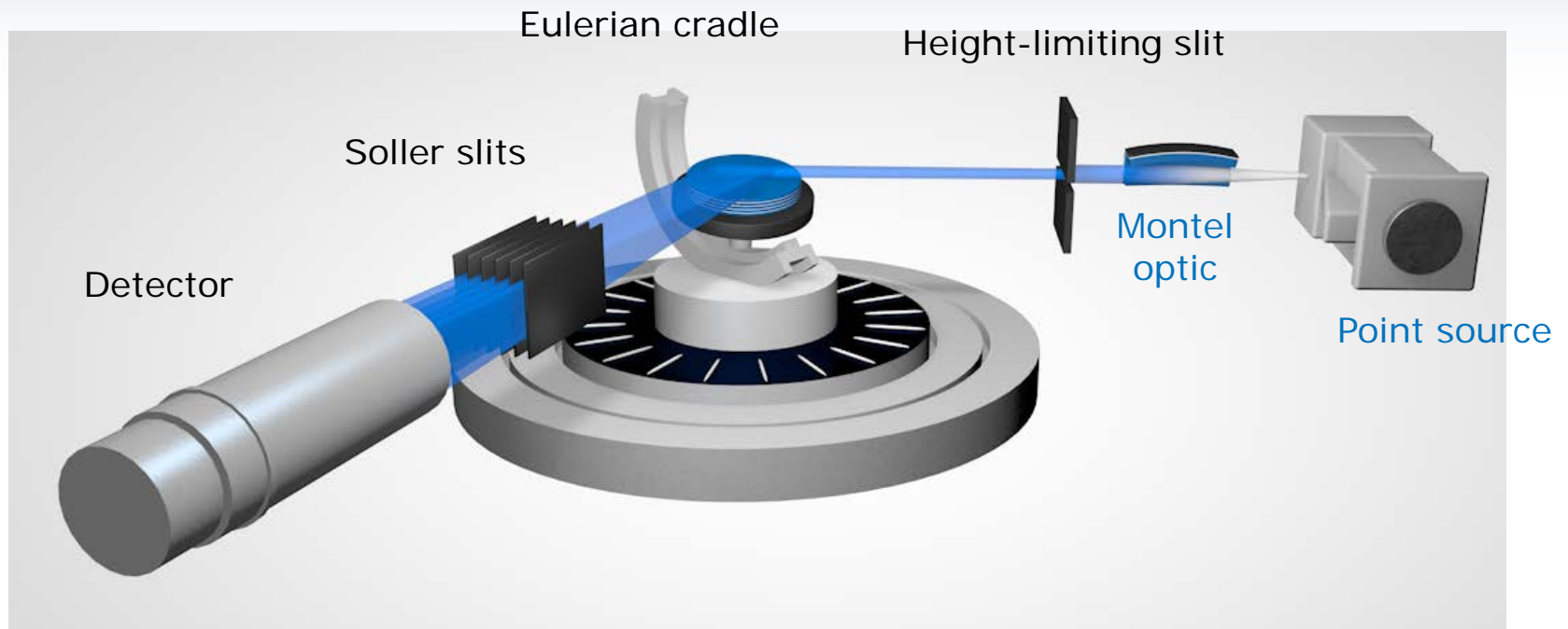


# Experimental setup for IPGID with PolyCapillary



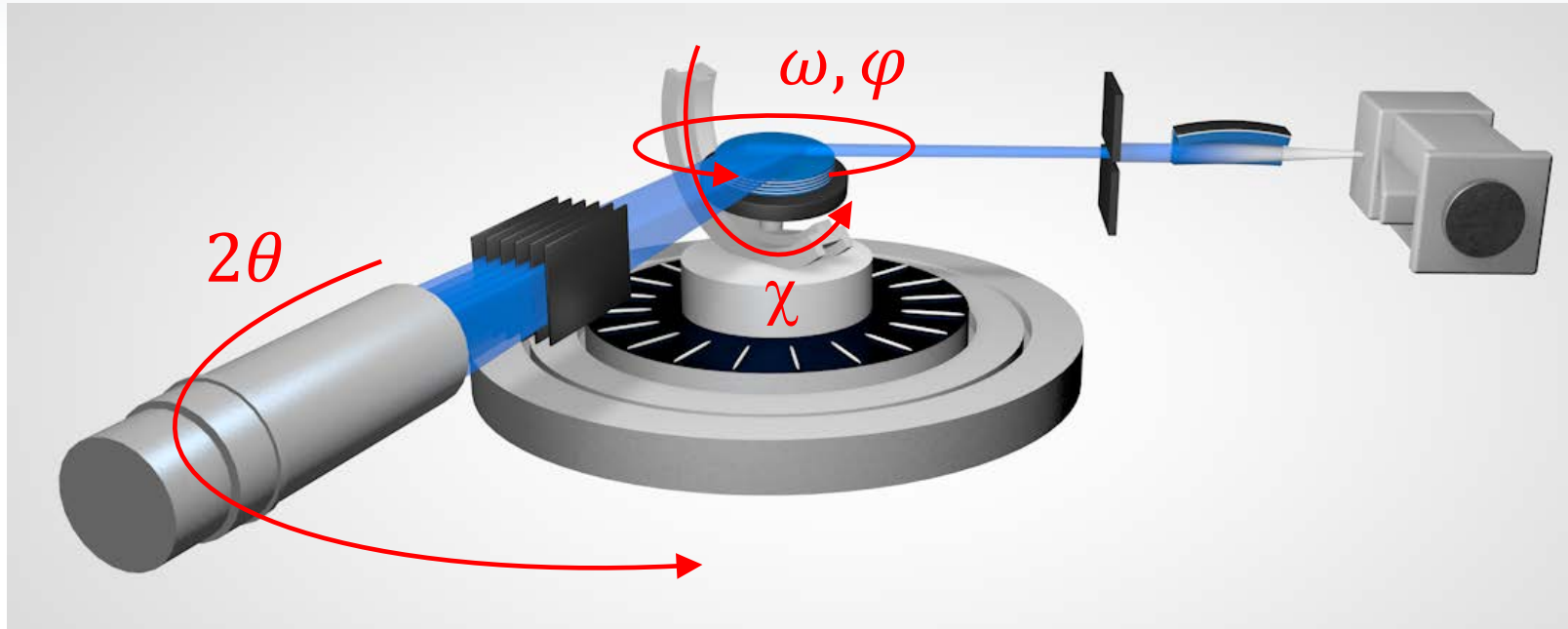
- PolyCap beam has poor resolution in surface-normal direction, no real depth control.
- Control of the incident angle via sample inclination.

# Experimental setup for IPGID with Montel optic



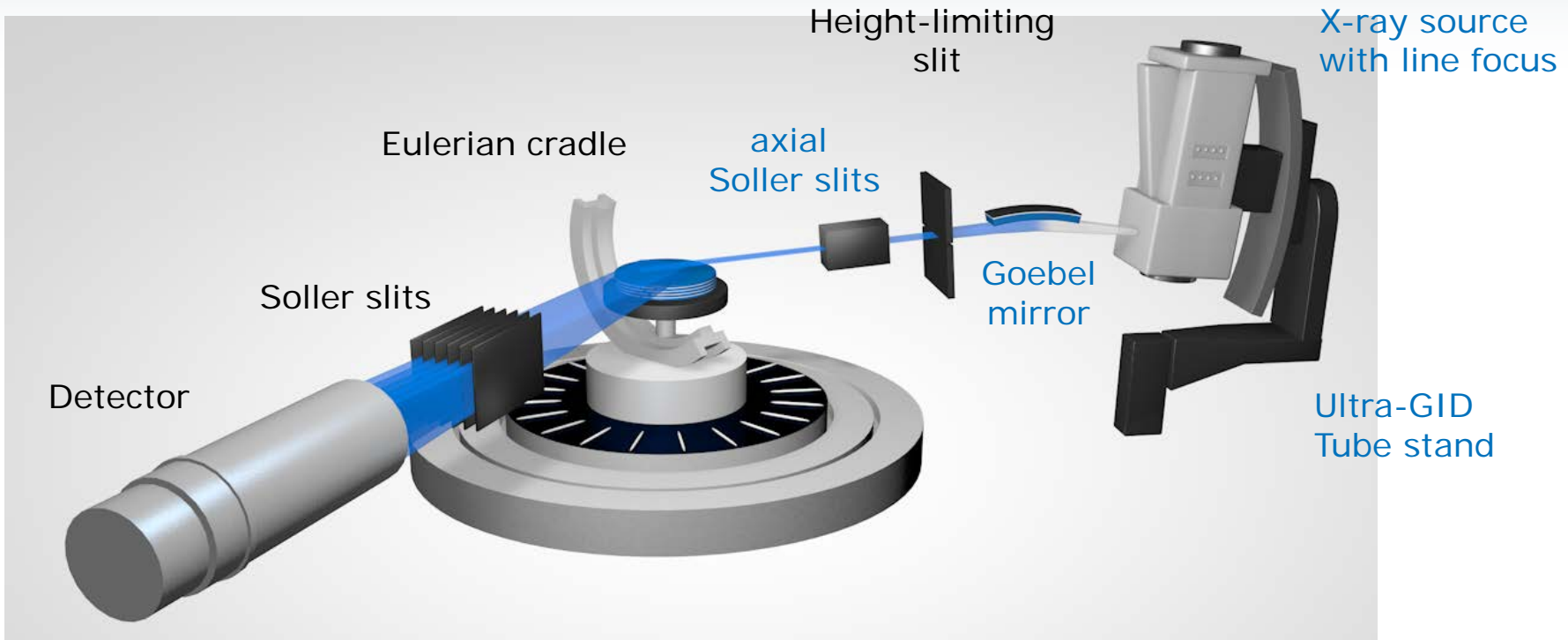
- Small incident beam (1 mm x 1 mm) with good in-plane resolution.
- Control of the incident angle via sample inclination.

# Experimental setup for IPGID with fixed point source



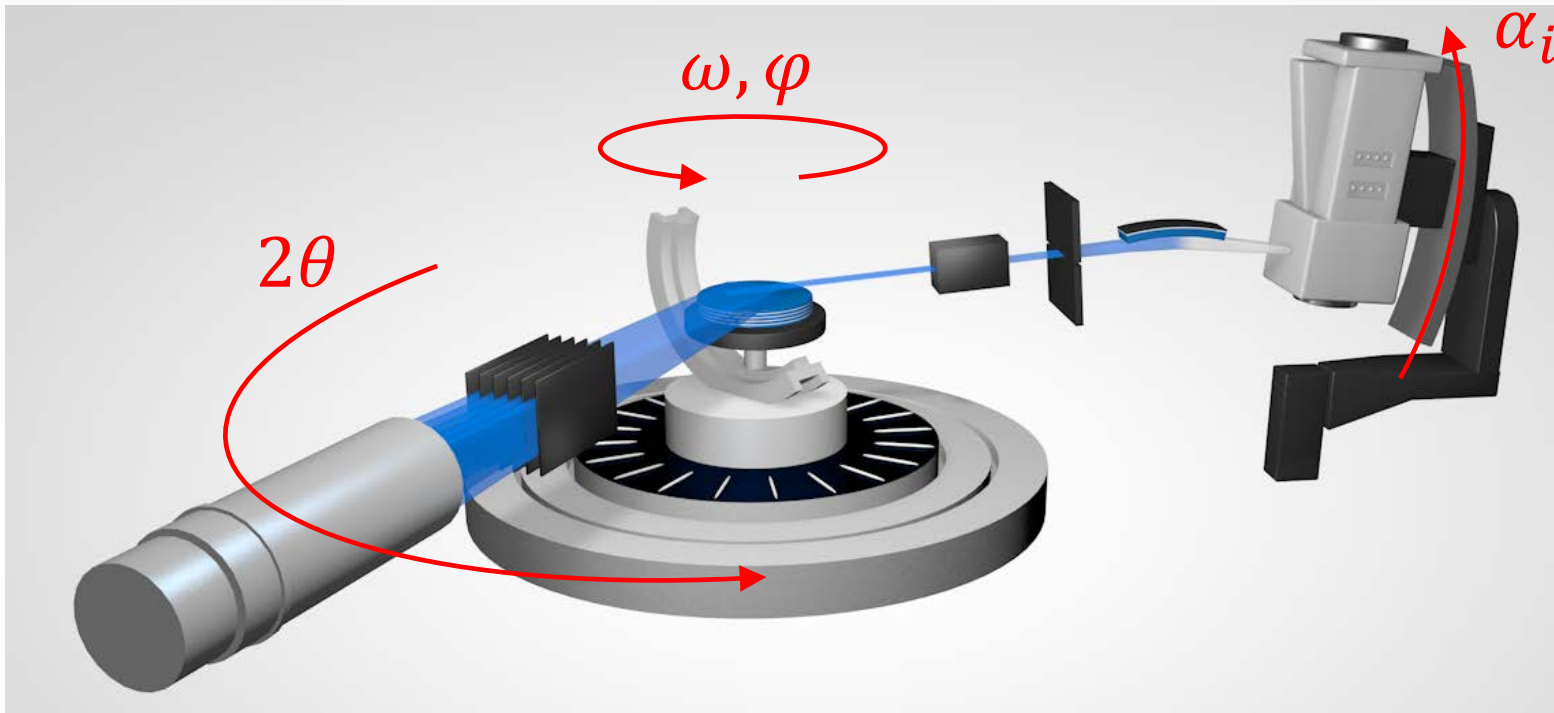
- Control of the incident angle via inclination of the sample using  $\chi$
- Not independent from  $\omega$

# Ultra-GID configuration : Optimized setup for surface diffraction



- Line focus is parallel to the sample surface: Good depth control.
- Angle of incidence is controlled by a separate drive.

# Ultra-GID configuration : Optimized setup for surface diffraction



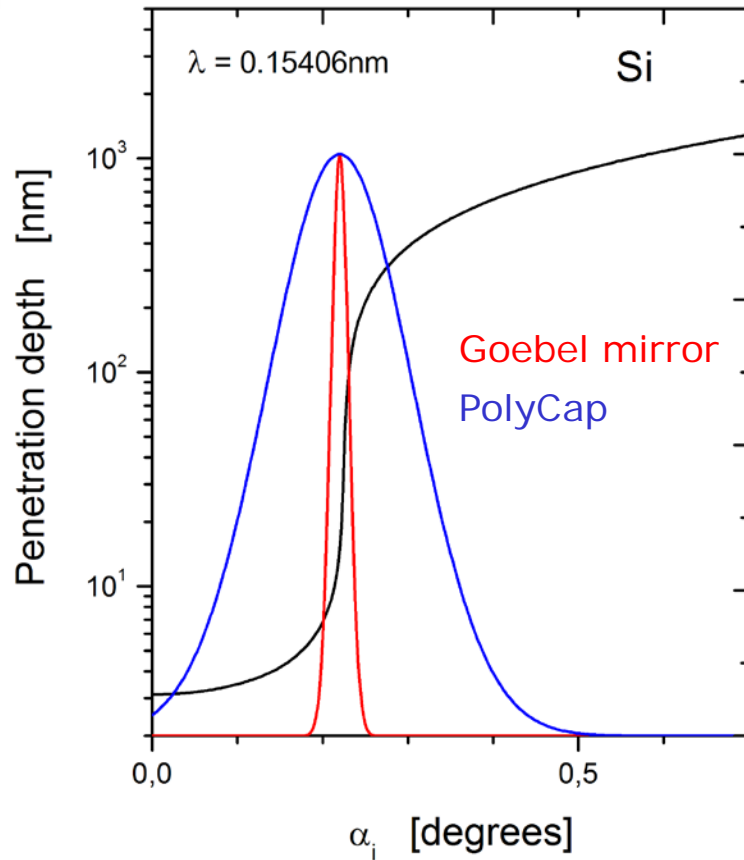
- Line focus is parallel to the sample surface: Good depth control.
- Angle of incidence is controlled by a separate drive.

# Preparing a beam with 200- $\mu\text{m}$ height: Comparison of the different setups

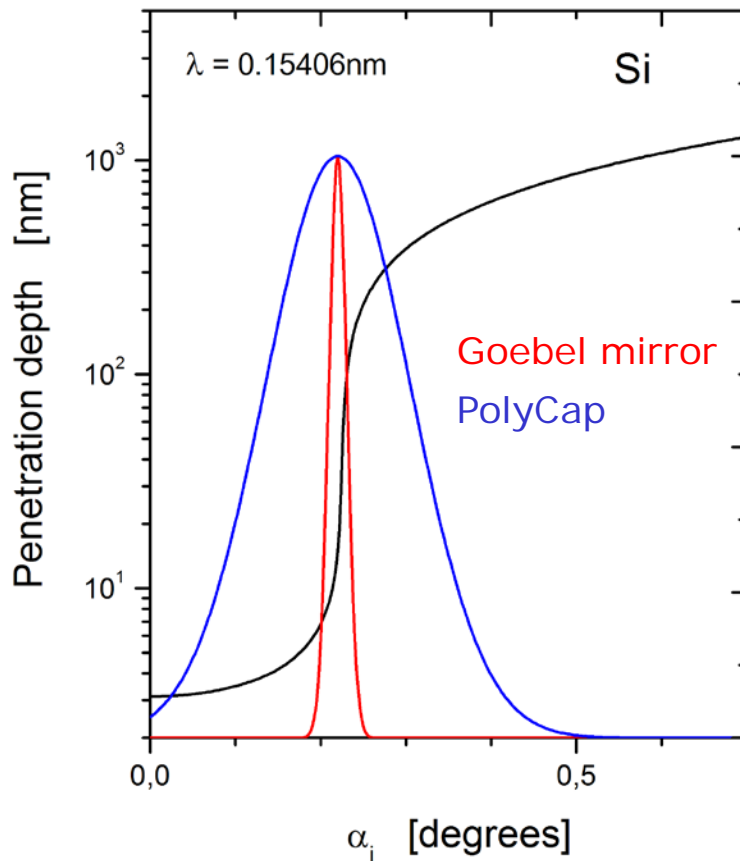


Optic	PolyCap	Montel	Ultra-GID with Goebel Mirror	
Focus orientation	Point focus	Point focus	Line focus	
Resolution $q_{\text{surface}}$	$0.2^\circ$	$0.05^\circ$	$0.023^\circ$	
Resolution $q_{\text{in-plane}}$	$0.2^\circ$	$0.05^\circ$	$0.2^\circ$ soller	$0.5^\circ$ soller
Beam width	5 cm	1 mm	16 mm	16 mm
Spectral purity	Tube spectrum	Cu $K\alpha_{1,2}$ , few $K\beta$	Cu $K\alpha_{1,2}$	

# Choice of the incident beam optics



# Choice of the incident beam optics



- The optimum choice of the incident beam configuration depends on the sample.
- **Depth control** requires good resolution perpendicular to the surface -> **Goebel mirror**
- Epitaxial samples with **low mosaicity** will except only small angular range of the incident beam -> **Goebel mirror**
- **Polycrystalline** samples and **thick layers** can be measured with a higher beam spread -> **PolyCap**

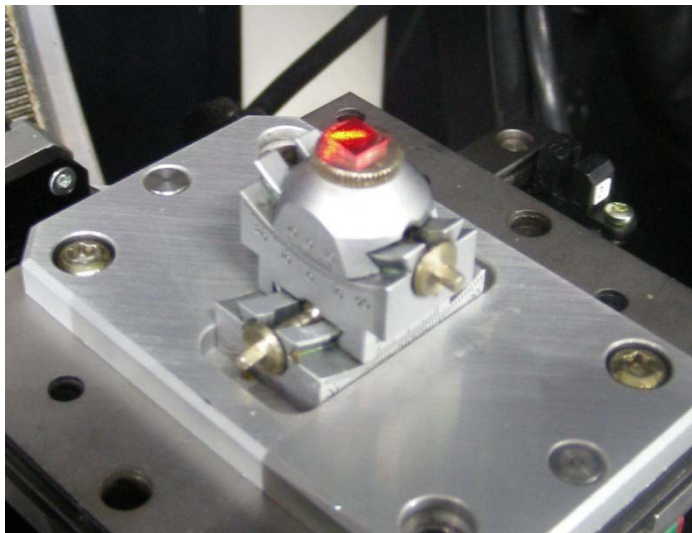


- Introduction
- Experimental configurations
- **Experimental tips**
- Examples

# Aligning the sample surface parallel to the $\phi$ -axis : tilt stages



- Manual goniometer head



- For small samples
- Optical alignment using a laser beam

- Motorized tilt stage

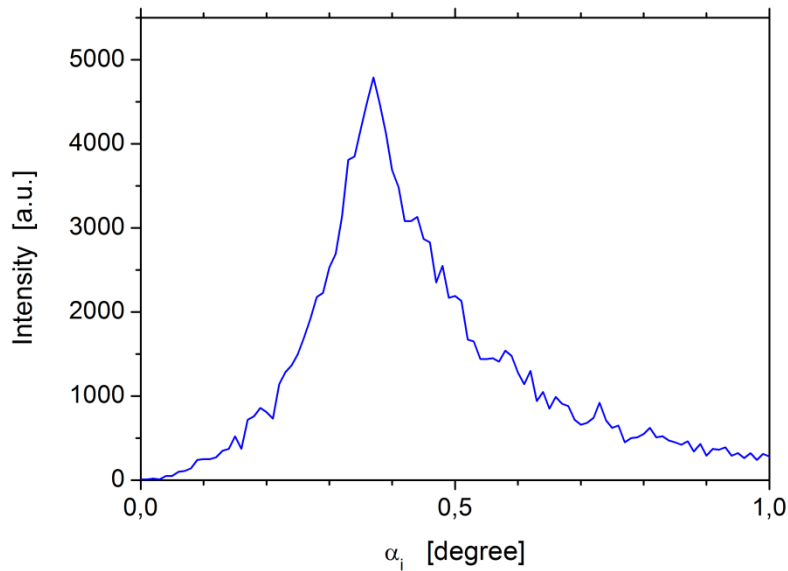


- For larger samples fixed by vacuum
- Can use the X-ray beam for surface alignment

# Optimizing the angle of incidence

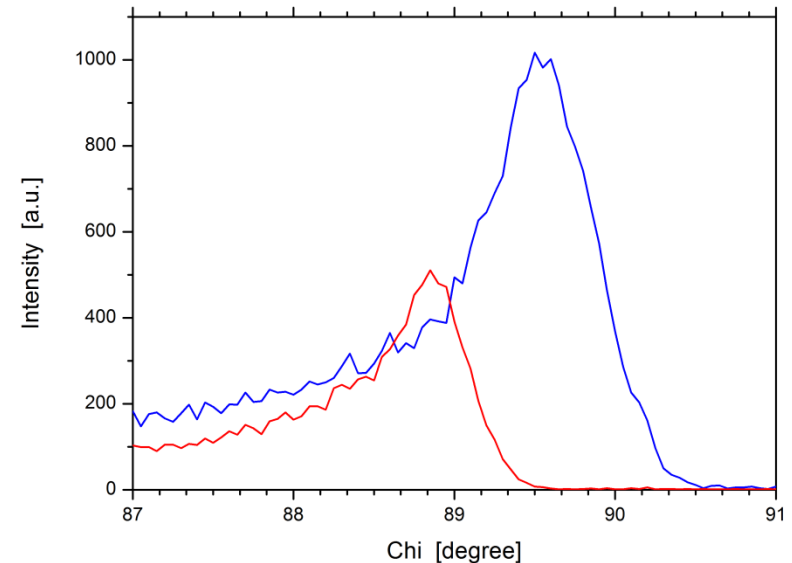


- With Ultra-GID tube stand



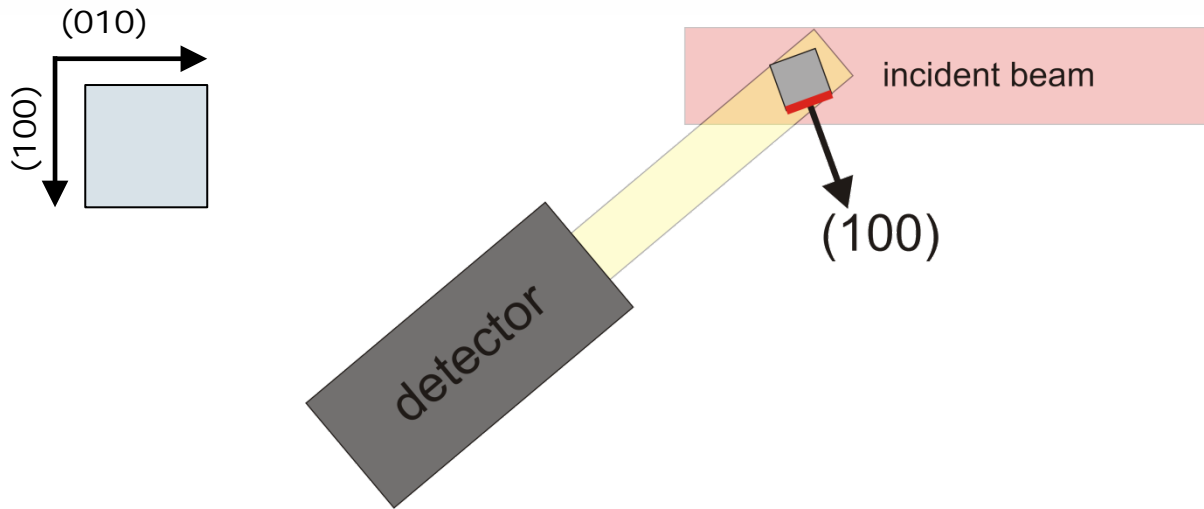
- Alignment of the optimal angle of incidence by  $\alpha_i$  drive.
- Independent of  $\omega$ .

- With fixed incident beam

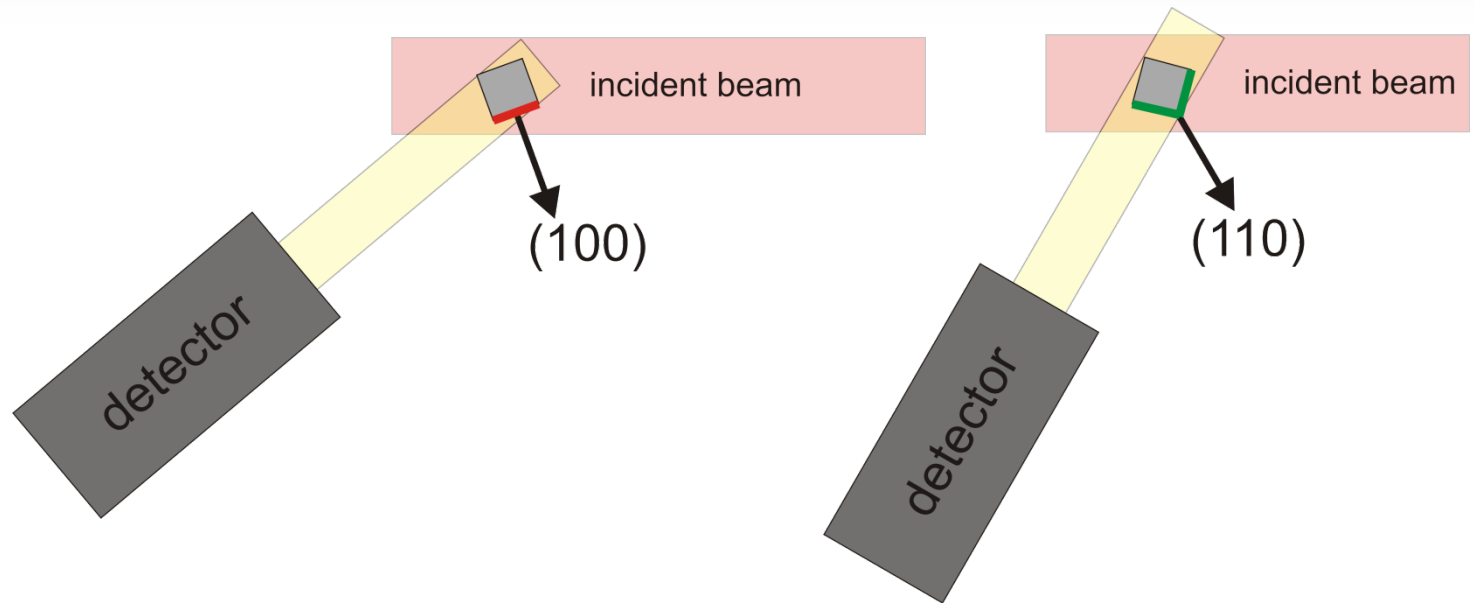
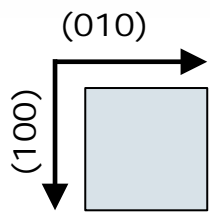


- Using  $\chi$  for inclining the sample surface.
- $\alpha_i = \cos(\chi) \omega$ . Depends on  $\omega$ .

# Choosing the appropriate reflection to avoid substrate scattering

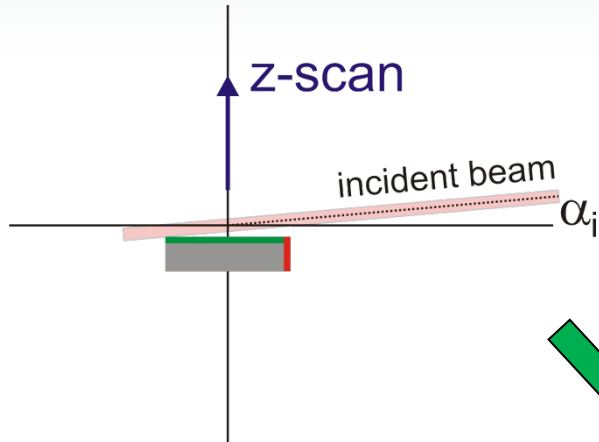


# Choosing the appropriate reflection to avoid substrate scattering

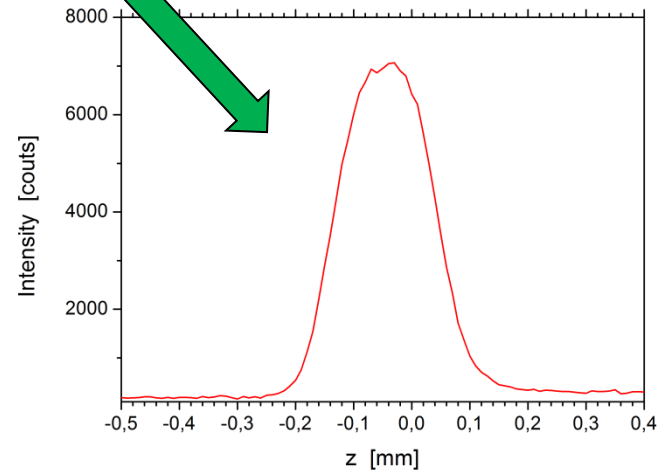


- Not the reflection with the highest intensity is always the best choice.
- Choosing an appropriate reflection helps to avoid scattering from the substrate edge.

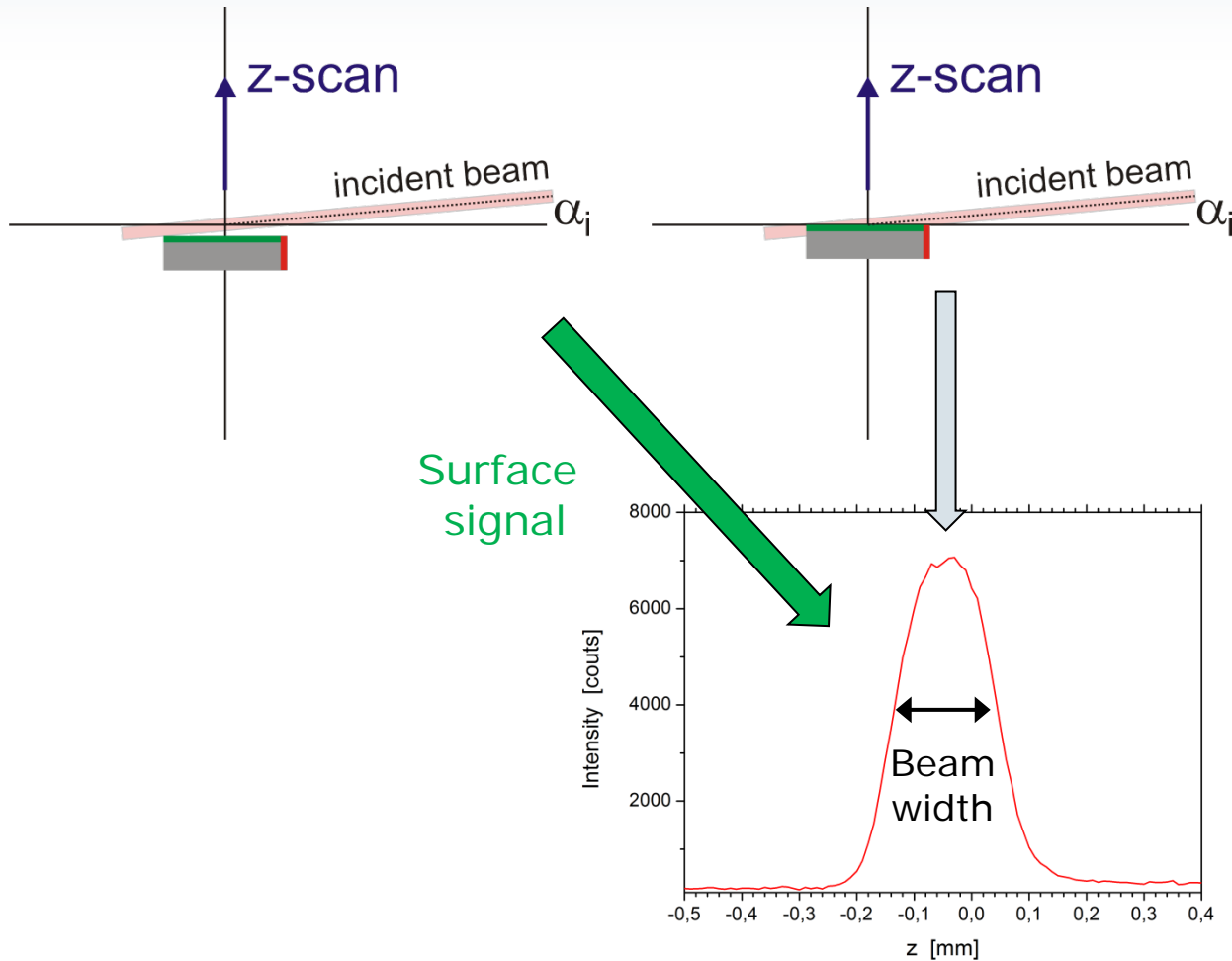
Alignment tips: Change **sample height** to avoid substrate scattering



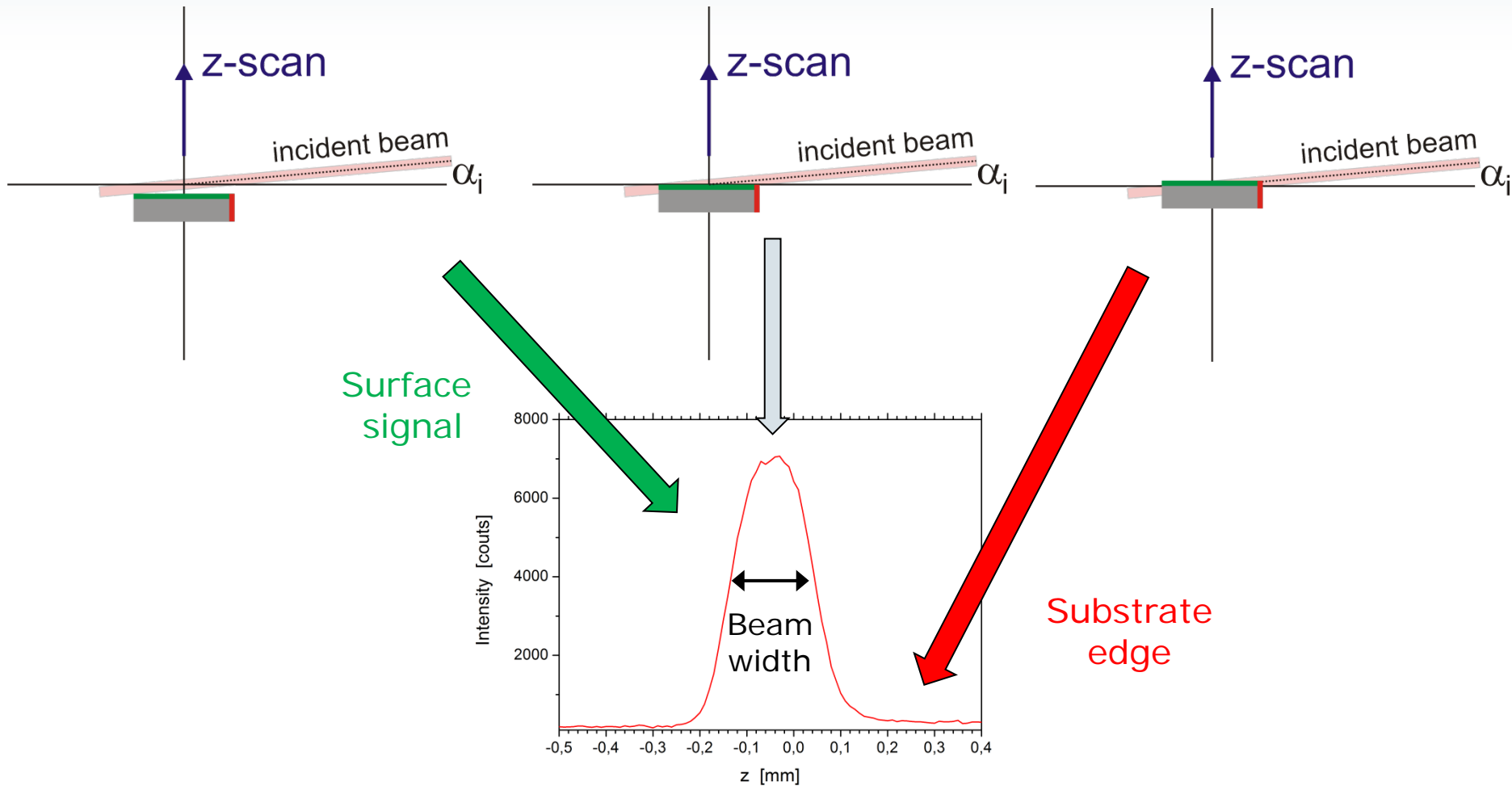
Surface  
signal



# Alignment tips: Change sample height to avoid substrate scattering

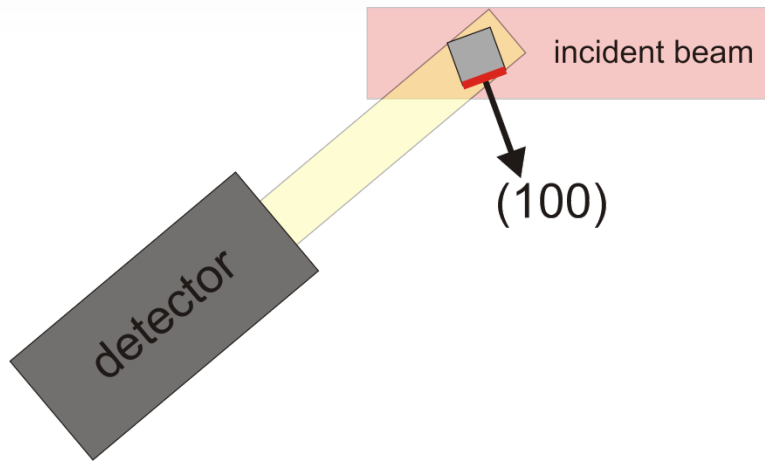


# Alignment tips: Change sample height to avoid substrate scattering

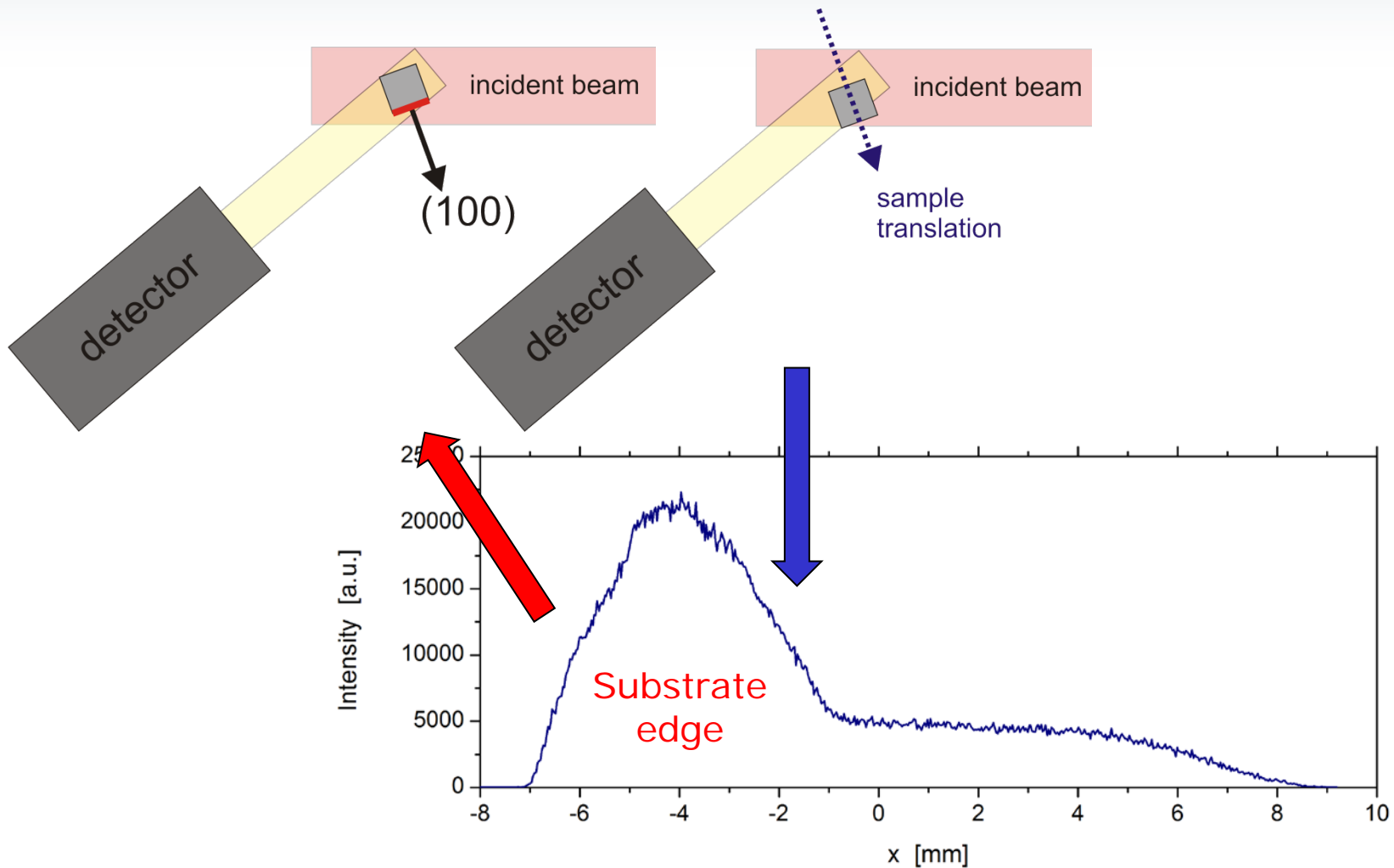




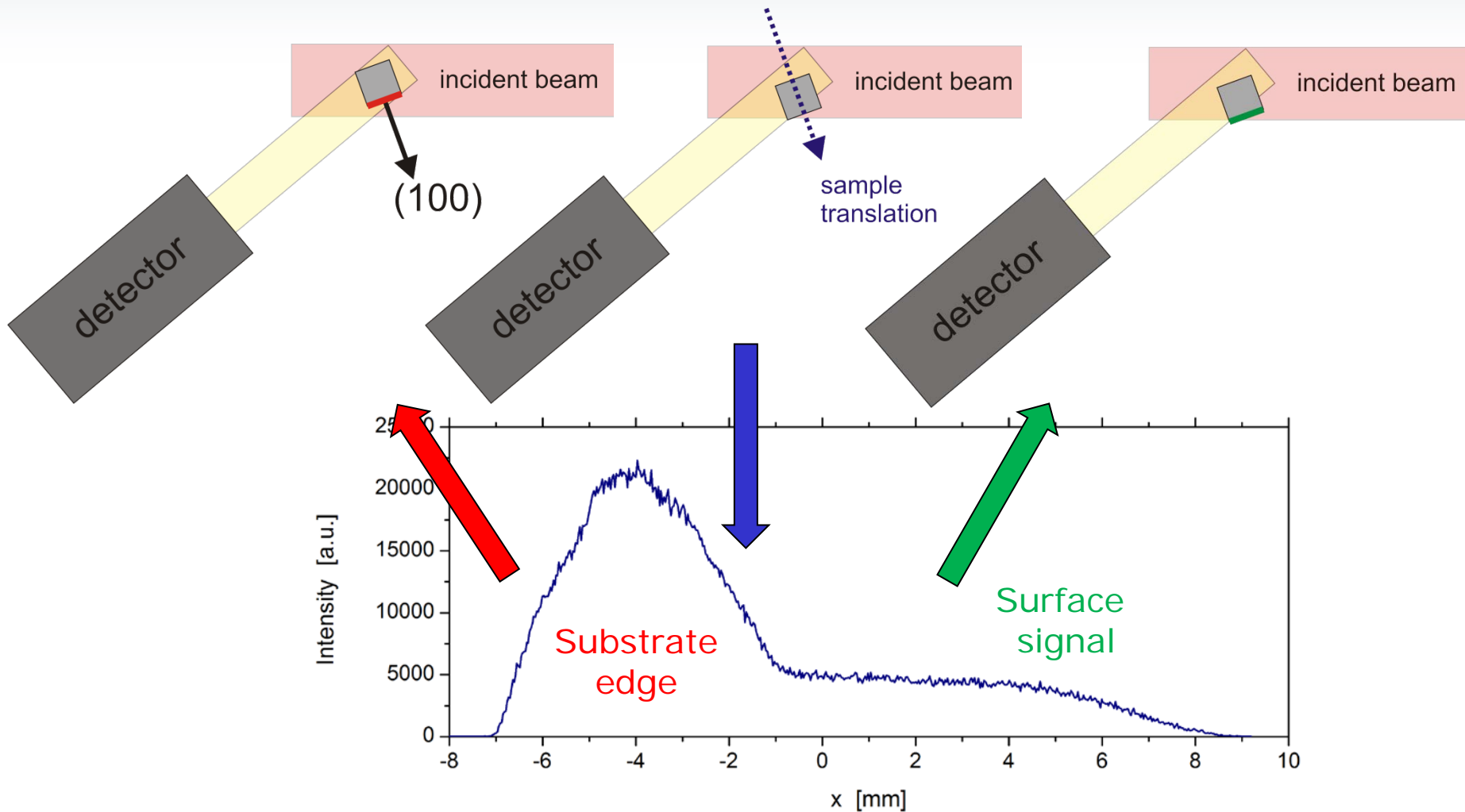
Alignment tips: **Sample translation** to avoid substrate scattering



# Alignment tips: Sample translation to avoid substrate scattering



# Alignment tips: Sample translation to avoid substrate scattering

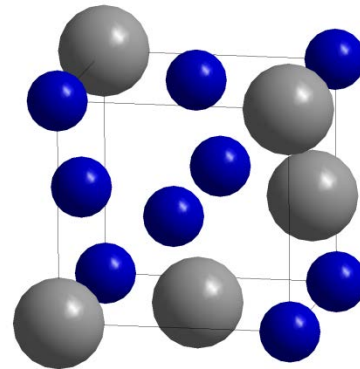


- Introduction
- Experimental configurations
- Experimental tips
- Examples
  - Polycrystalline samples
  - Epitaxial grown samples
  - In-plane diffraction with 1D detector

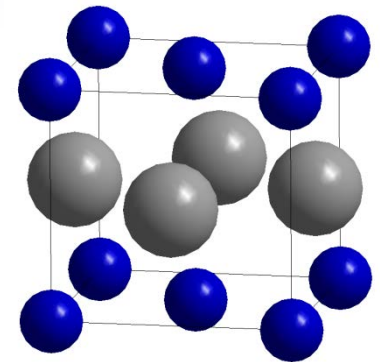
# Structure determination of polycrystalline FePt thin films



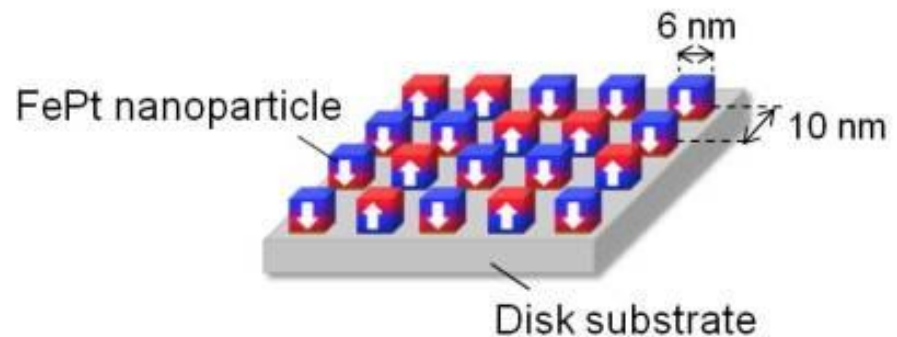
- FePt is promising material for magnetic mass storage devices
- A1 phase (face-centered cubic)
- L1<sub>0</sub> phase (face-centered tetragonal) ferromagnetic
- Thickness of the FePt film: 10 nm



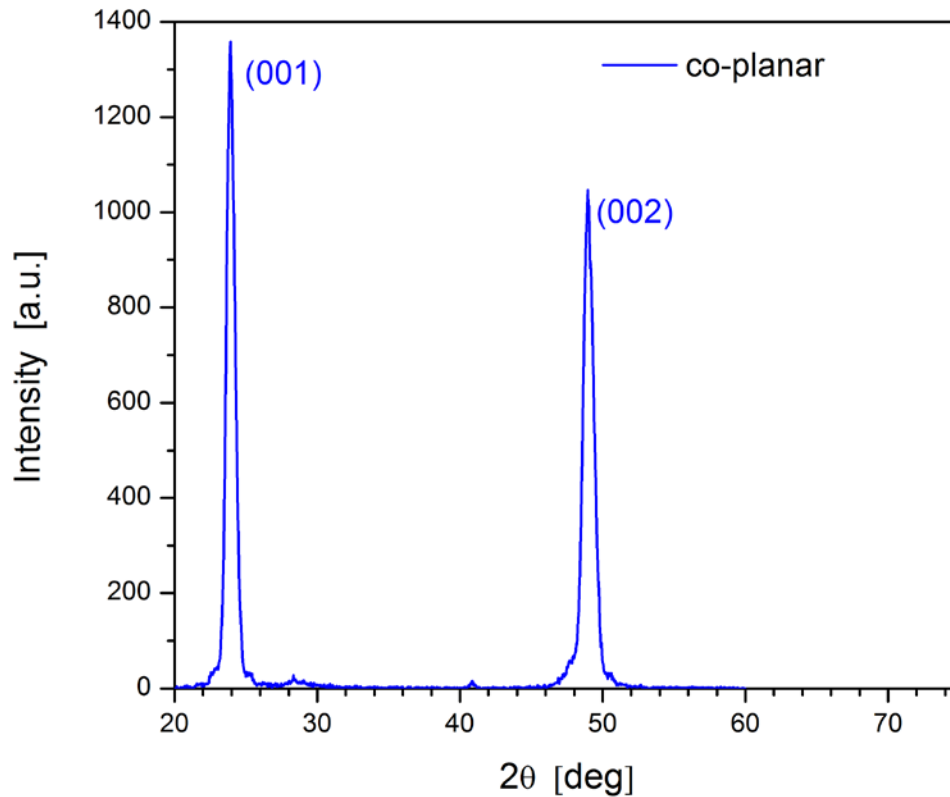
FePt - A1 phase



FePt - L1<sub>0</sub> phase

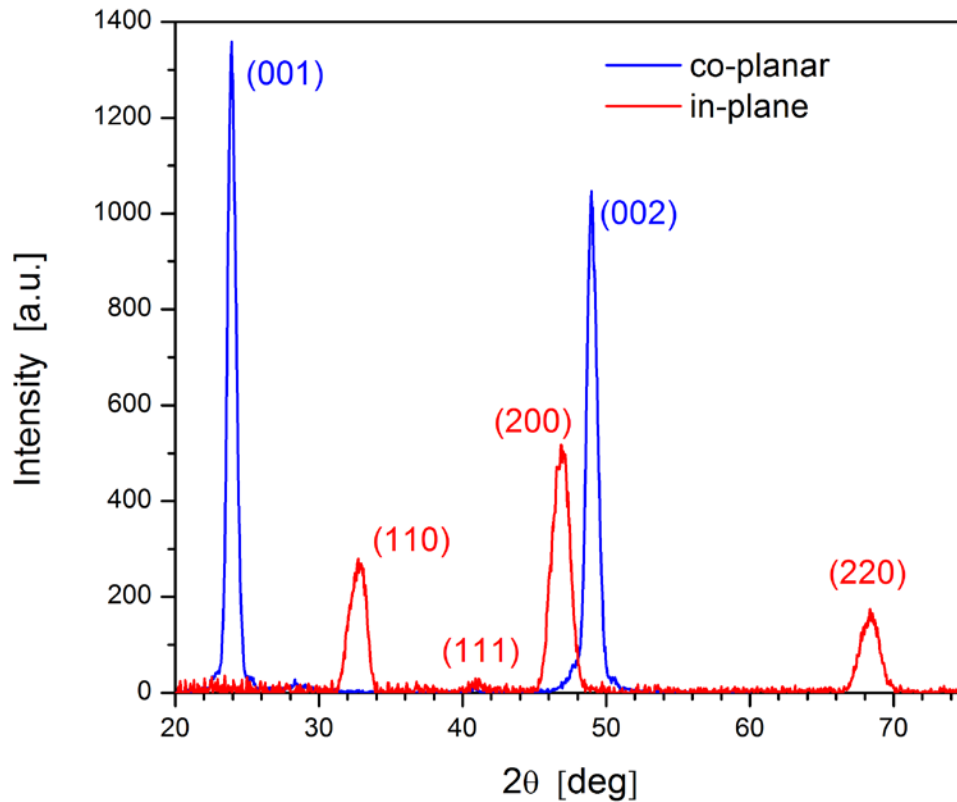


# Structure determination of polycrystalline FePt thin films



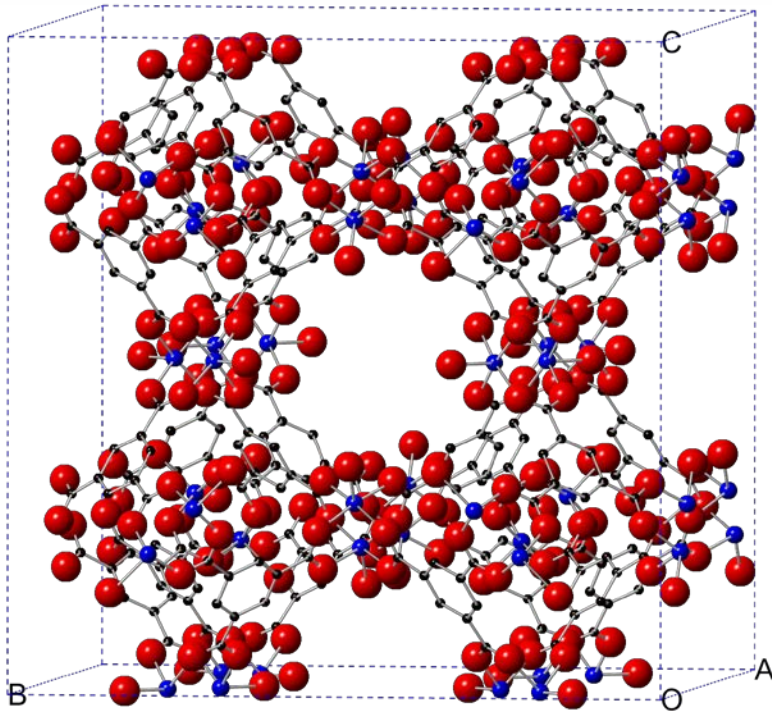
- Crystallite size in the surface normal direction is about the film thickness.

# Structure determination of polycrystalline FePt thin films



- Crystallite size in the surface normal direction is about the film thickness.
- In-plane crystallite size is about 6.5 nm
- In-plane fiber textured around (001)

# In-plane GID on Metal-organic frameworks (MOF's)

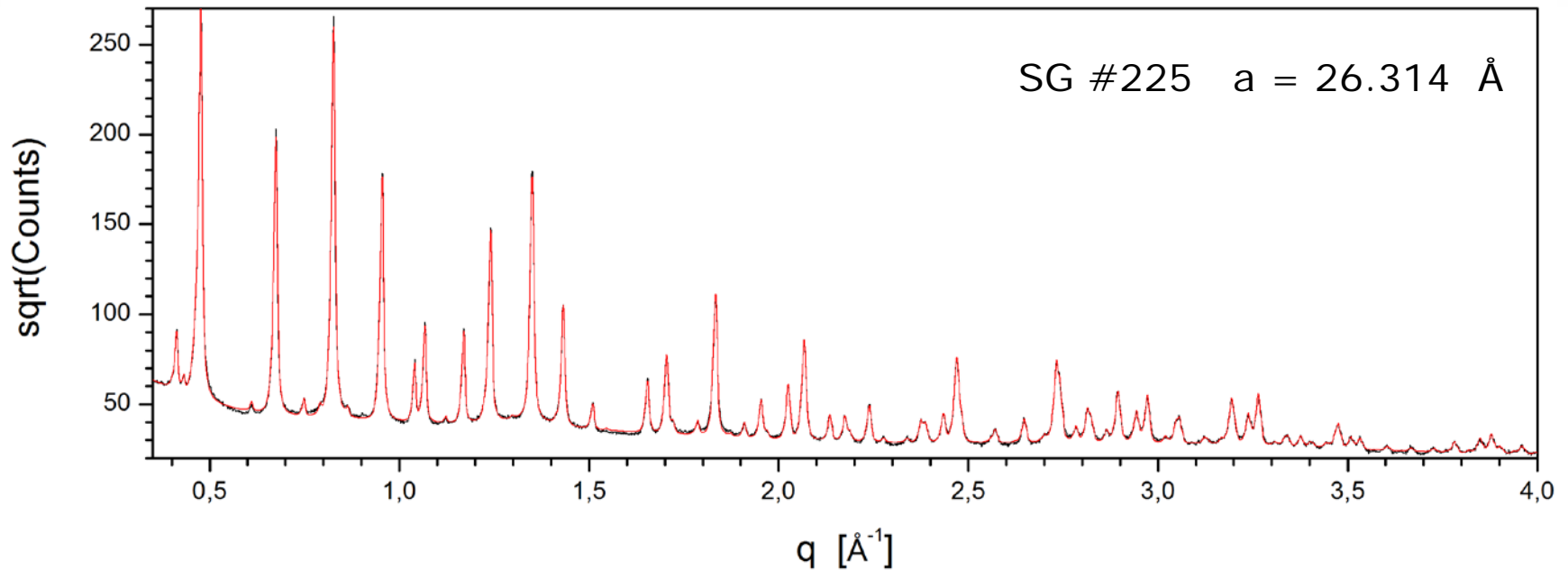


- Topic of current research
- Incorporation of nanoparticles, e.g.  $\text{Au}_9$  or  $\text{Au}_{55}$  : controlling refractive index
- Drug carrier and release systems
- HKUST-1 :  $\text{C}_{18}\text{H}_6\text{Cu}_3\text{O}_{12}$
- Space group 225 with lattice constant  $a = 26.314 \text{ \AA}$ .

MOF samples kindly provided by P. Weidler, IFG, KIT / Germany

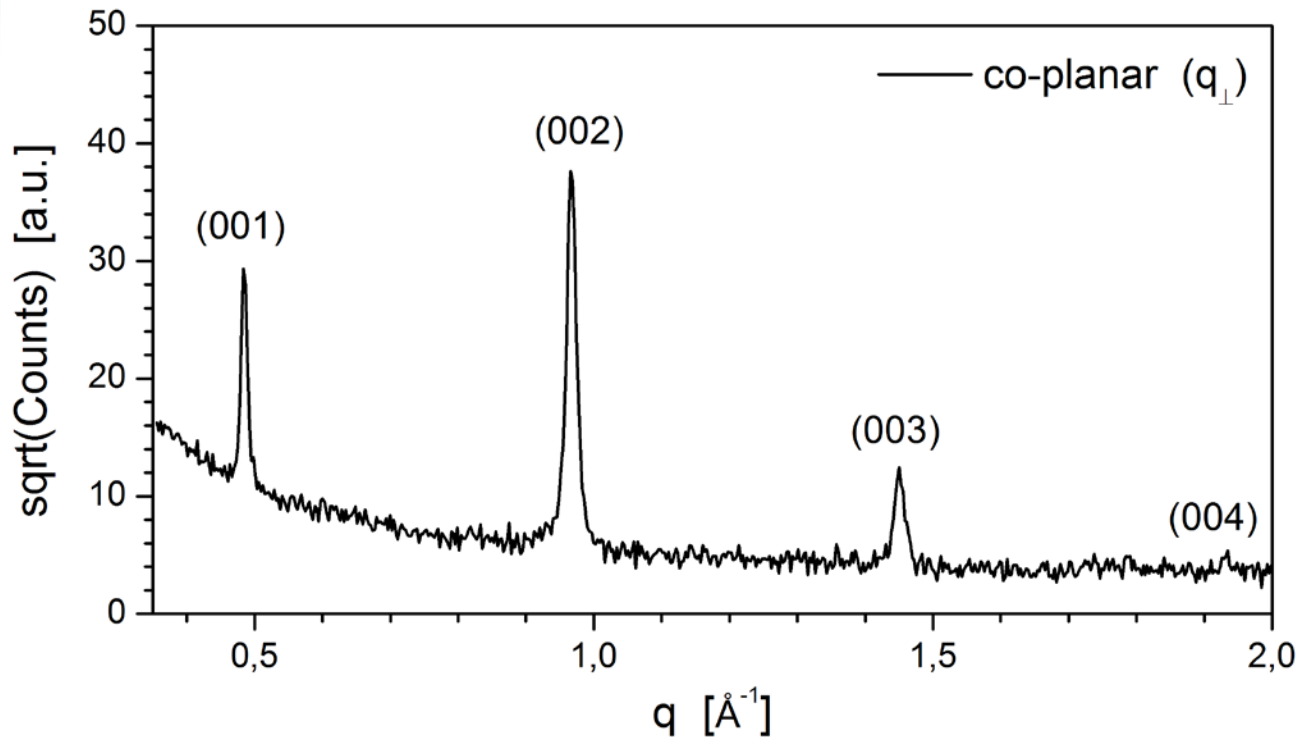


# In-plane GID on Metal-organic frameworks (MOF's)



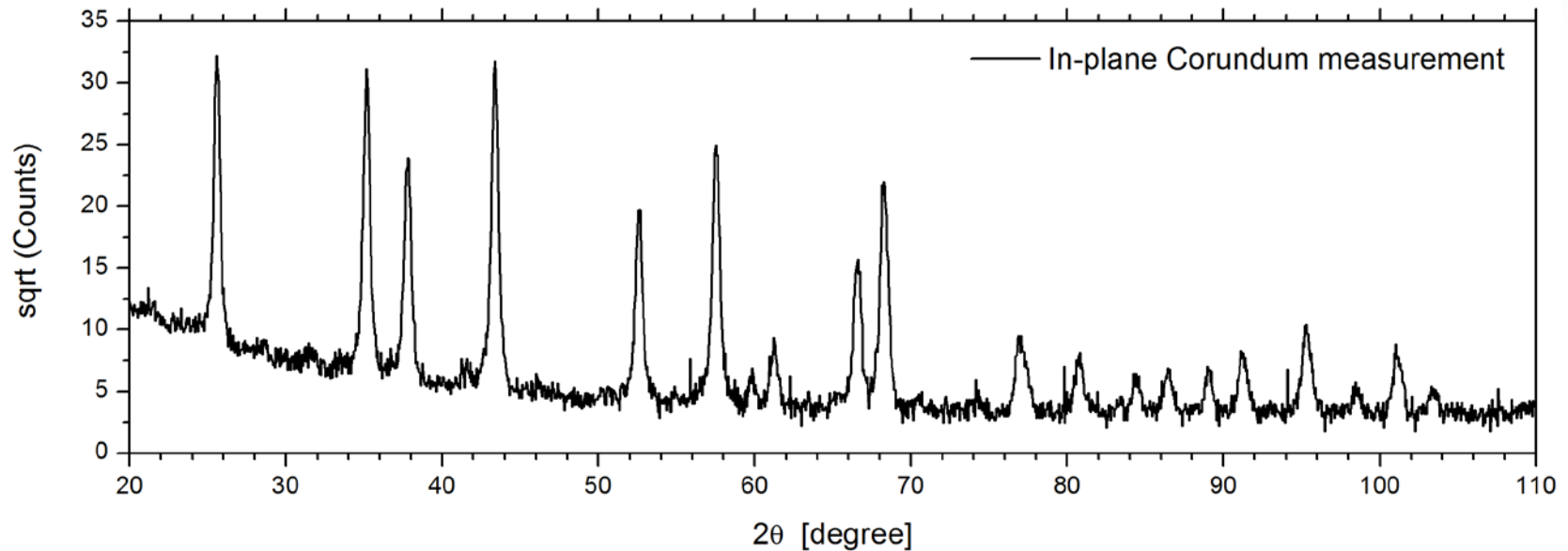
- Measurement of HKUST-1 powder provides structure information.
- Crystallite size is about 195 nm.

# In-plane GID on Metal-organic frameworks (MOF's)



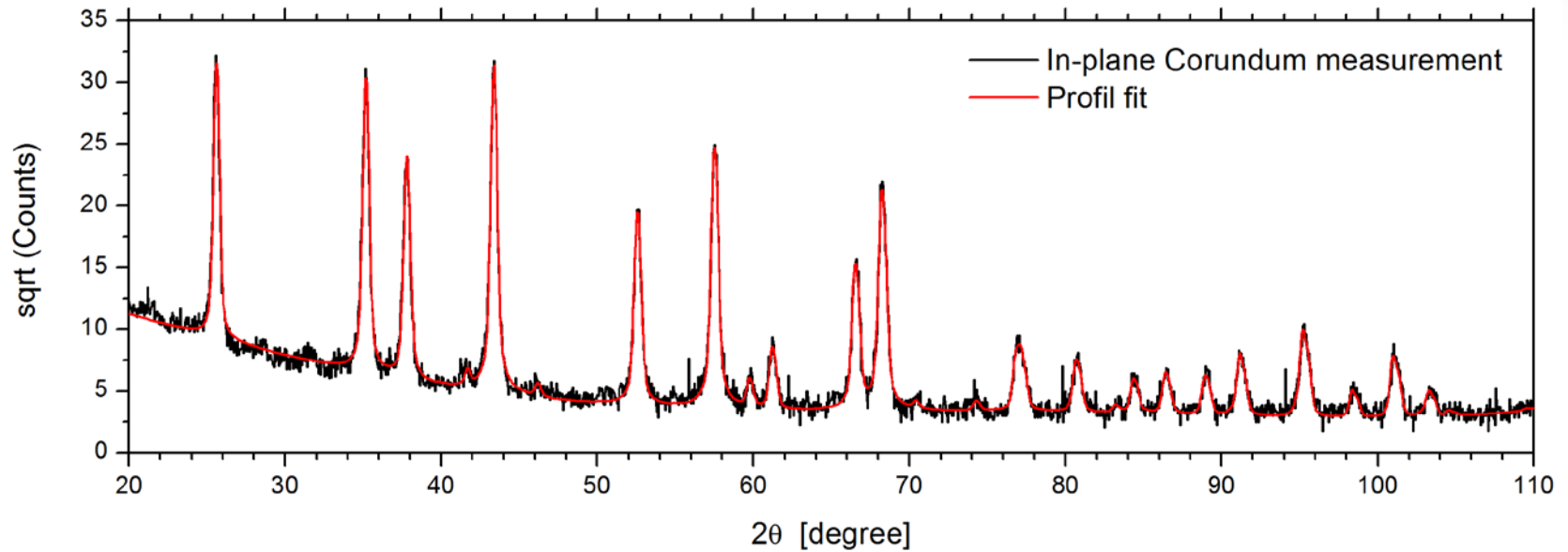
- MOF crystallites with (001) orientation and size of about 90 nm.

# Determination of the [in-plane resolution function](#)



- Precise determination of the crystallite size requires knowledge of the [resolution function](#) for the used experimental setup.
- Use polycrystalline sample with high crystallite size, e.g. NIST SRM 1976 (Corundum).

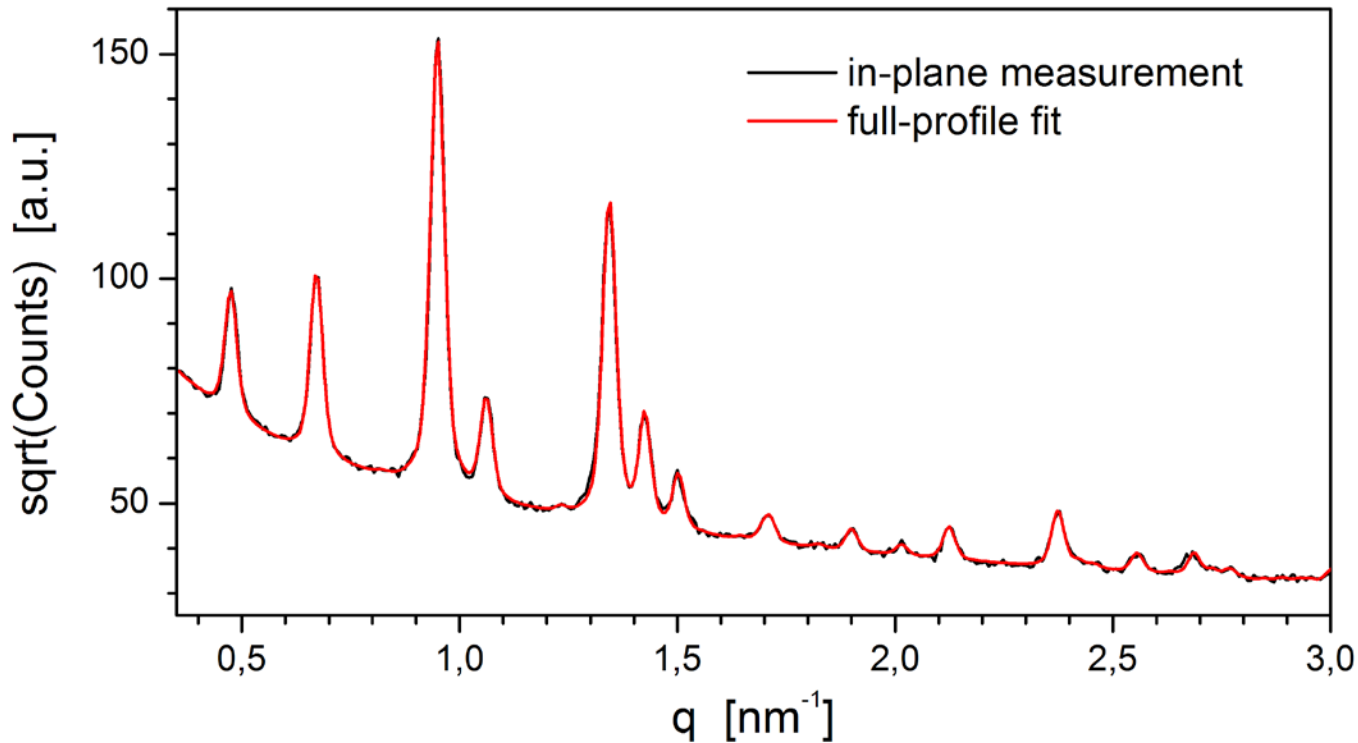
# Determination of the in-plane resolution function



- Full profile fit provides the resolution function.

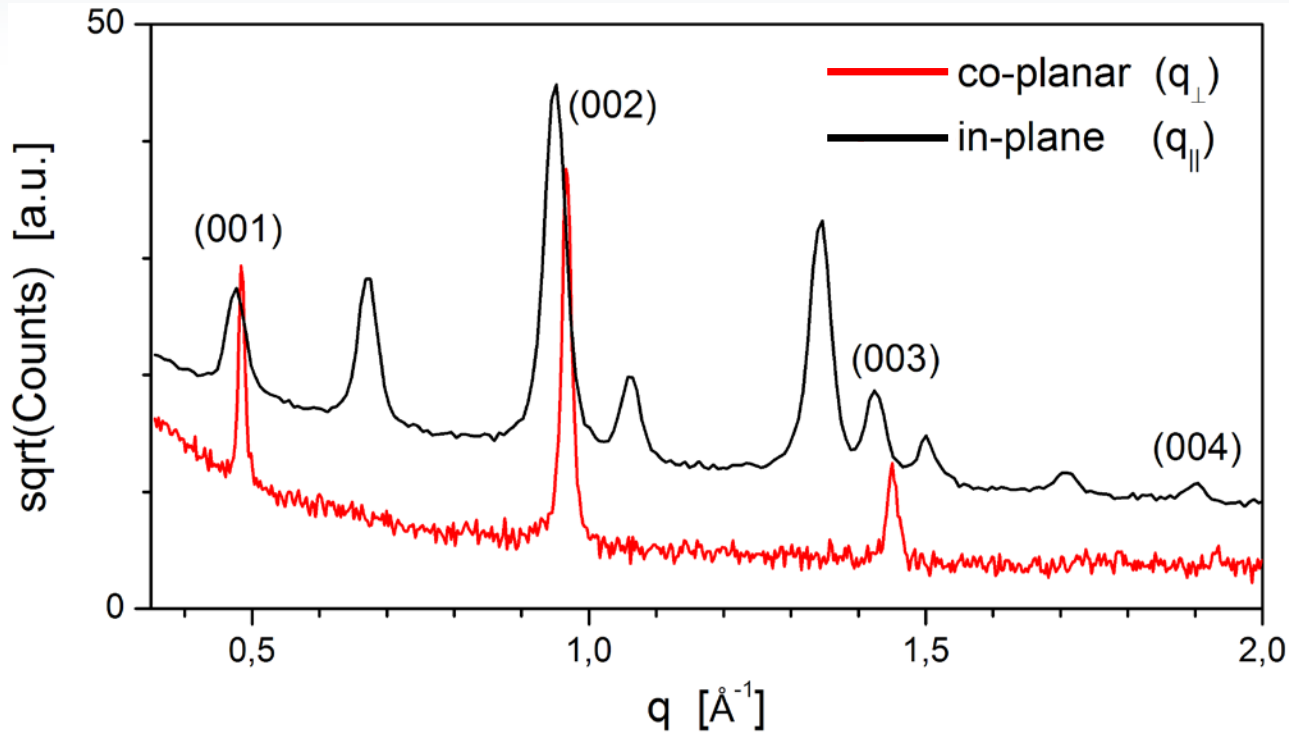
	Conv. Type	2Th Dependence	Use	Value	Code
1	Lorentzian	1/Cos(Th)	<input checked="" type="checkbox"/>	0.02052447	Refine
2	Gaussian	1/Cos(Th)	<input checked="" type="checkbox"/>	0.03920406	Refine
3	Lorentzian	Constant	<input checked="" type="checkbox"/>	0.1060961	Refine
4	Gaussian	Constant	<input checked="" type="checkbox"/>	0.2940303	Refine

# Determination of the in-plane crystallite size



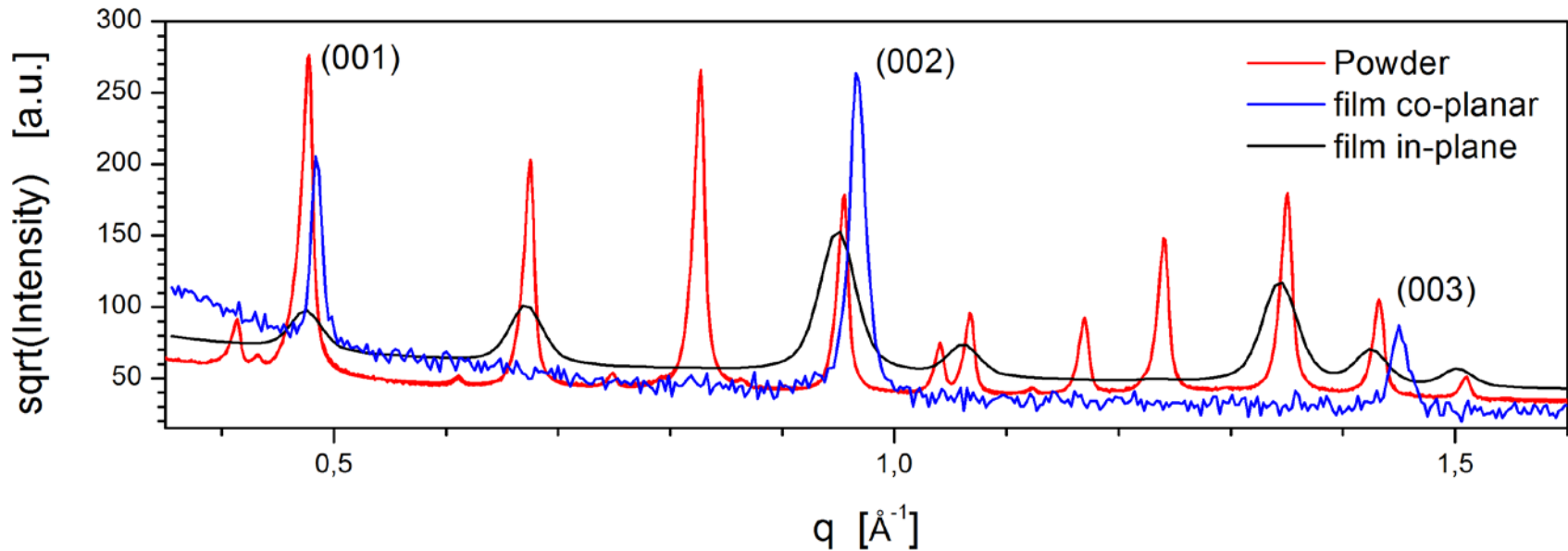
- Use the known resolution function
- Full profile fit yields **120-nm crystallite size** parallel to the surface.

# In-plane GID on Metal-organic frameworks (MOF's): **crystallite size**



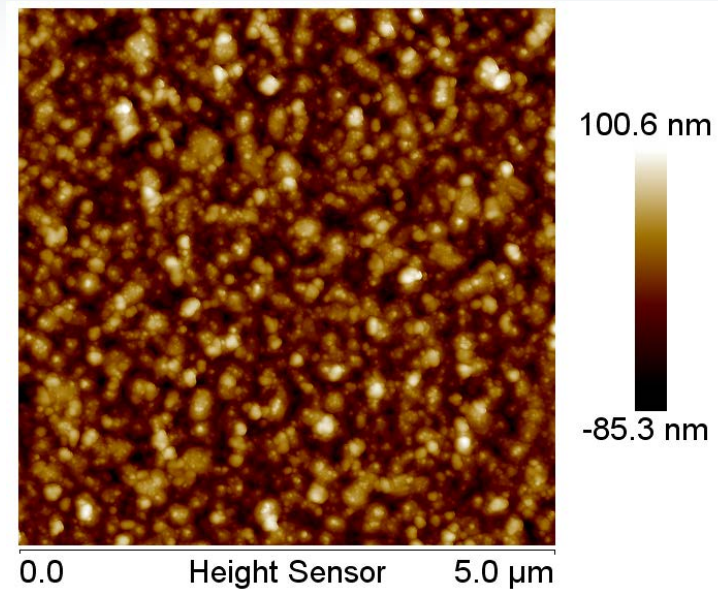
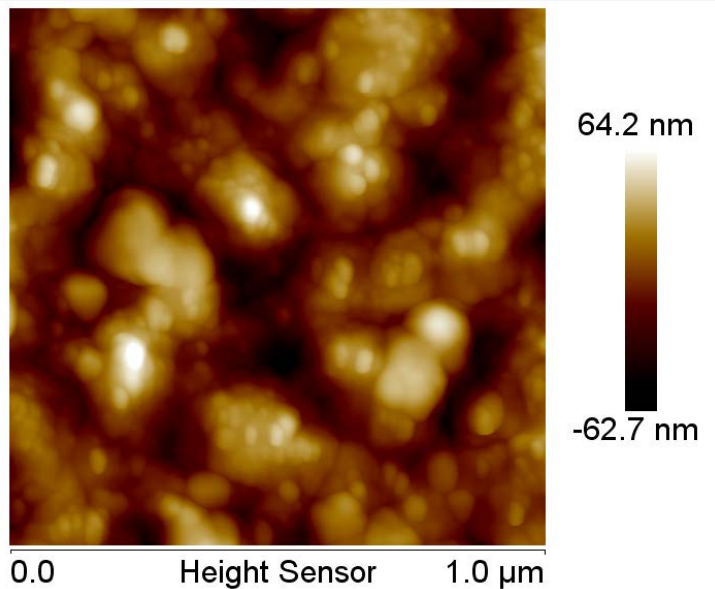
- MOF crystallites with (001) surface normal and size of about 90 nm.
- Fiber textured with 120 nm crystallite size parallel to the surface.

# In-plane GID on Metal-organic frameworks (MOF's): lattice constants

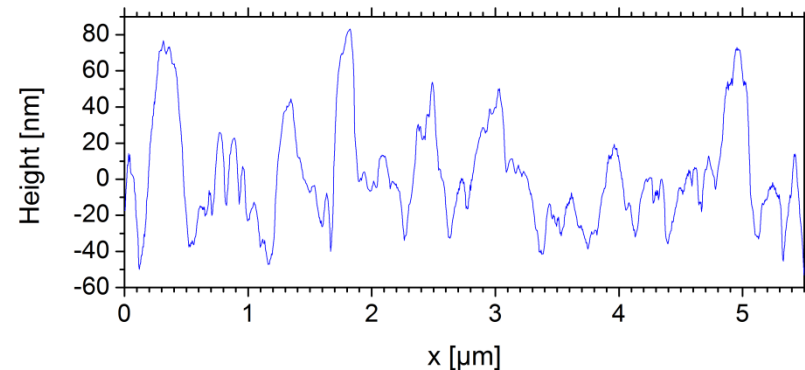


- MOF crystallites with (001) surface normal and size of about 90 nm.
- In-plane lattice parameter: 26.482  $\text{\AA}$  Crystallite size about 120  $\text{\AA}$
- Co-planar lattice parameter: 26.0055  $\text{\AA}$

# In-plane GID on Metal-organic frameworks (MOF's): AFM pictures



- The AFM pictures yield particles with size of 250-350 nm.
- This is not the crystallite size.





- Introduction
- Experimental configurations
- Performing an experiment
- Examples
  - Polycrystalline samples
  - Epitaxial grown samples
  - In-plane diffraction with 1D detector

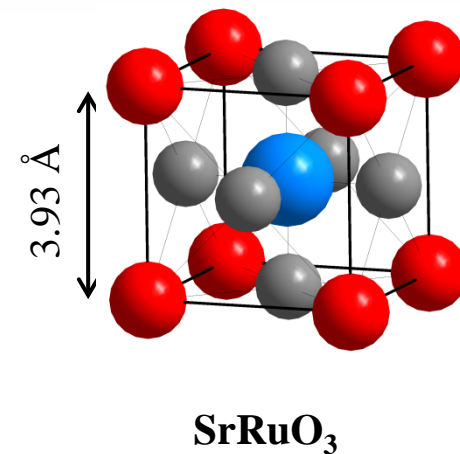
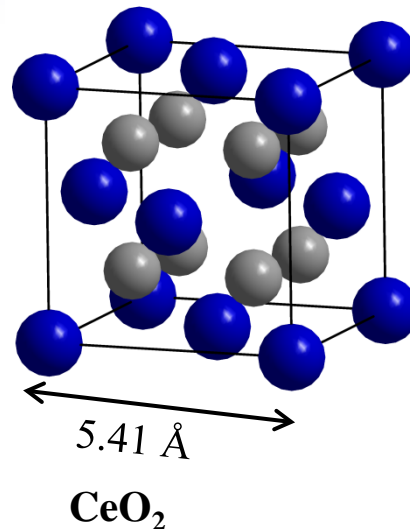
# Application 3

## Probing in-plane symmetry directly



40nm SrRuO <sub>3</sub>
40nm CeO <sub>2</sub>
substrate Al <sub>2</sub> O <sub>3</sub>

By Madhana Sunder,  
Bruker AXS, Madison(WI) \*



- Determine the epitaxial relationship.
- Based on lattice mismatch one would expect the unit cells to exhibit a twisted cube on cube epitaxy.

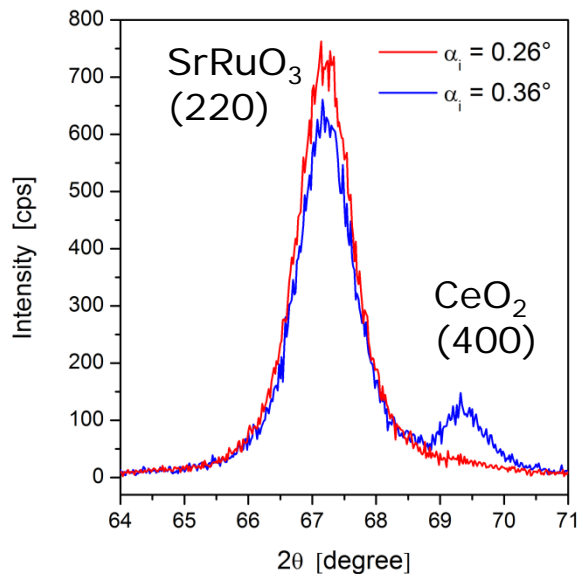
\* *Growth of heteroepitaxial single crystal Lead Magnesium Niobate-Lead Titanate thin films on r-plane Sapphire substrates*, Doctoral dissertation, **Madhana Sunder**, 2009

# Application 3

## Probing in-plane symmetry directly



- $2\theta/\omega$ -scan at  $\text{SrRuO}_3$  (220)



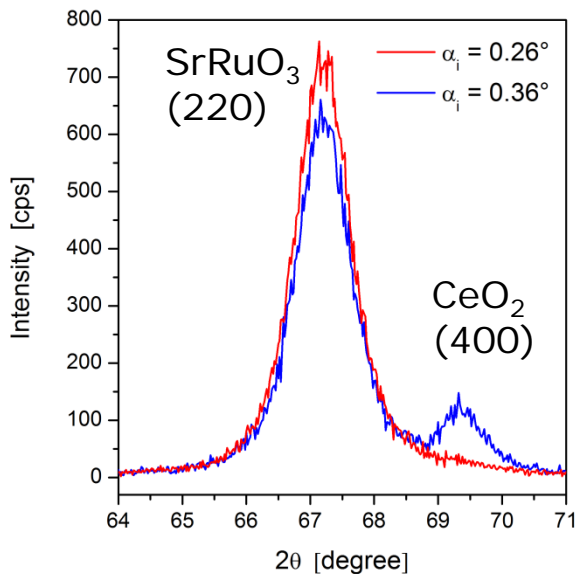
- $\text{SrRuO}_3$  (220) ||  $\text{CeO}_2$  (100)
- Clear isolation of  $\text{SrRuO}_3$  (220) reflection by depth control

# Application 3

## Probing in-plane symmetry directly

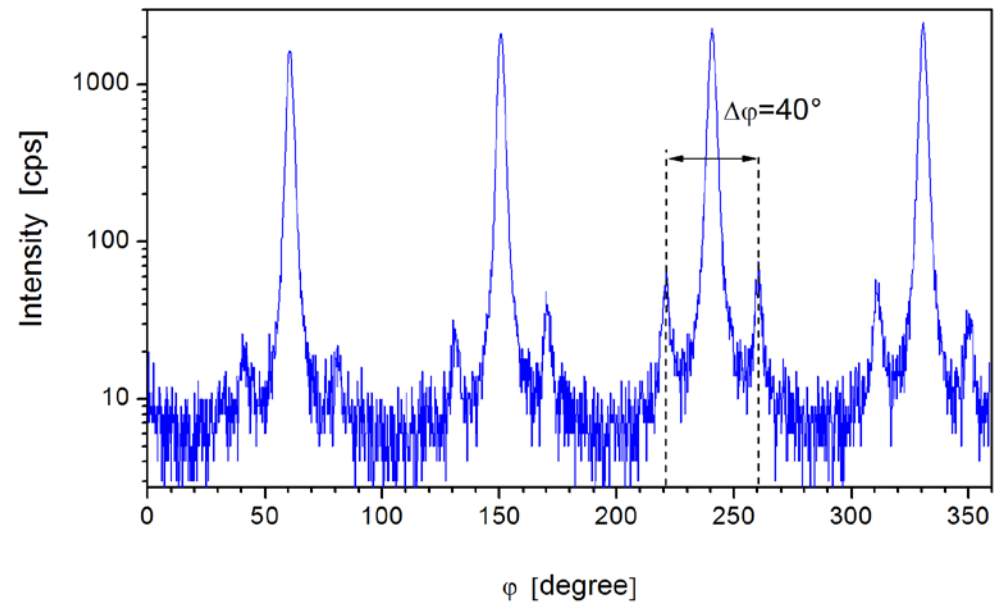


- $2\theta/\omega$ -scan at  $\text{SrRuO}_3$  (220)



- $\text{SrRuO}_3$  (220) ||  $\text{CeO}_2$  (100)
- Clear isolation of  $\text{SrRuO}_3$  (220) reflection by depth control

- $\phi$ -scan at  $2\theta$  of  $\text{SrRuO}_3$  (220)



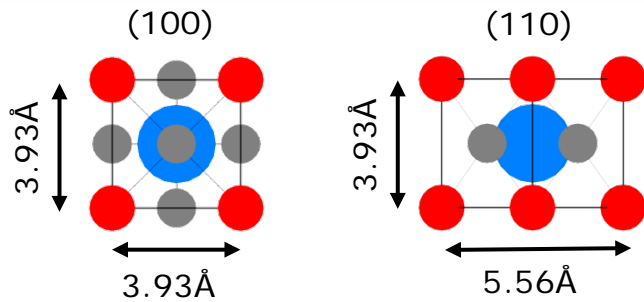
- A simple rotation of the sample around the surface normal directly reveals the in-plane symmetry.
- Requires surface normal ||  $\phi$ -axis.

# Application 3

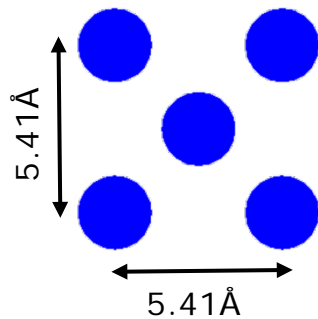
## Probing in-plane symmetry directly



$\text{SrRuO}_3$

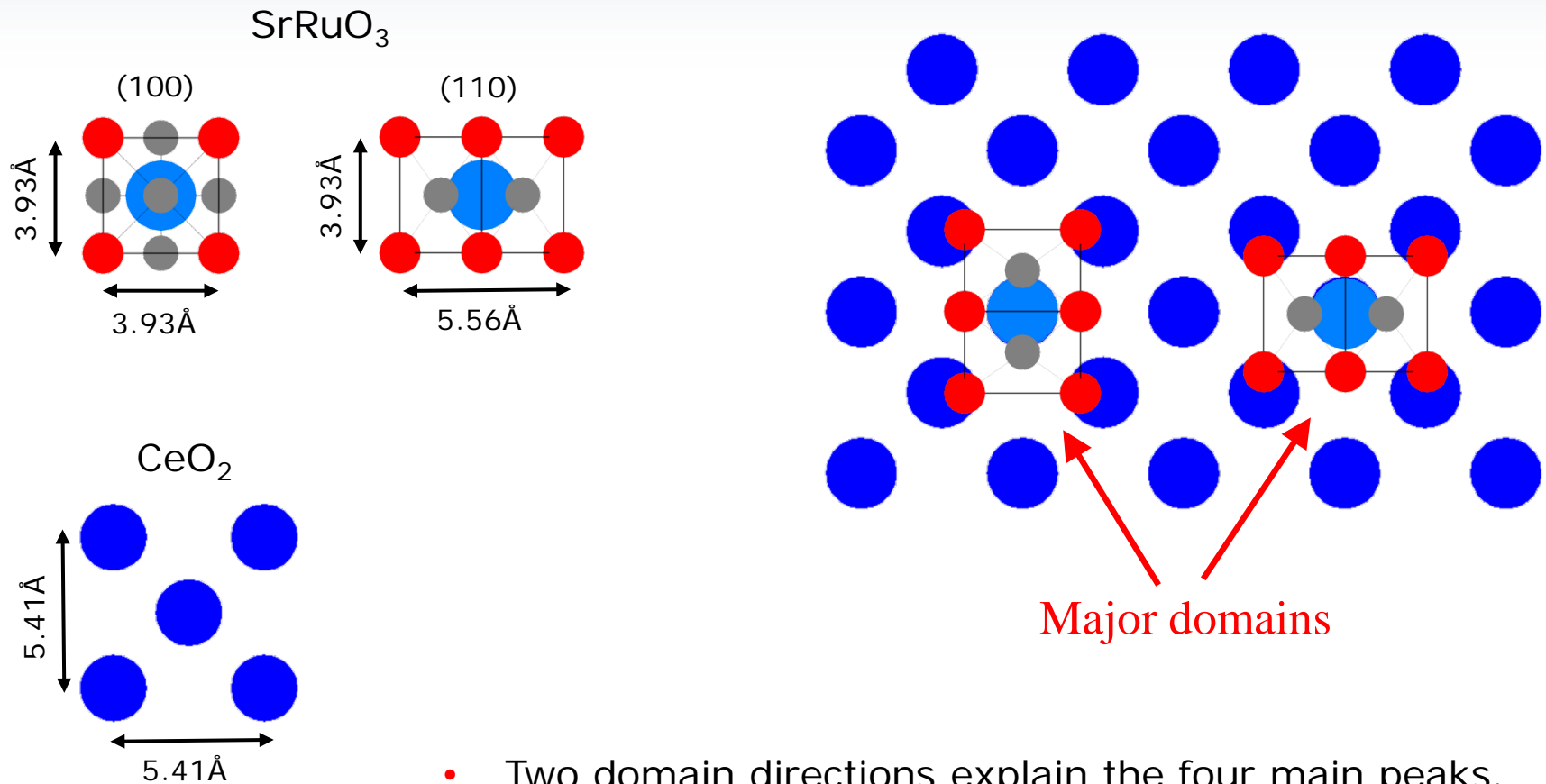


$\text{CeO}_2$



# Application 3

## Probing in-plane symmetry directly



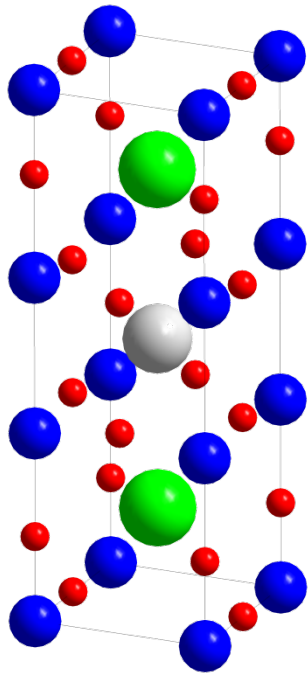
- Two domain directions explain the four main peaks.
- Smaller satellite peaks indicate additional domains.

# YBCO on STO

## Determination of epitaxial relations

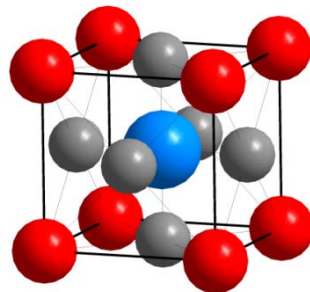


YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7</sub>

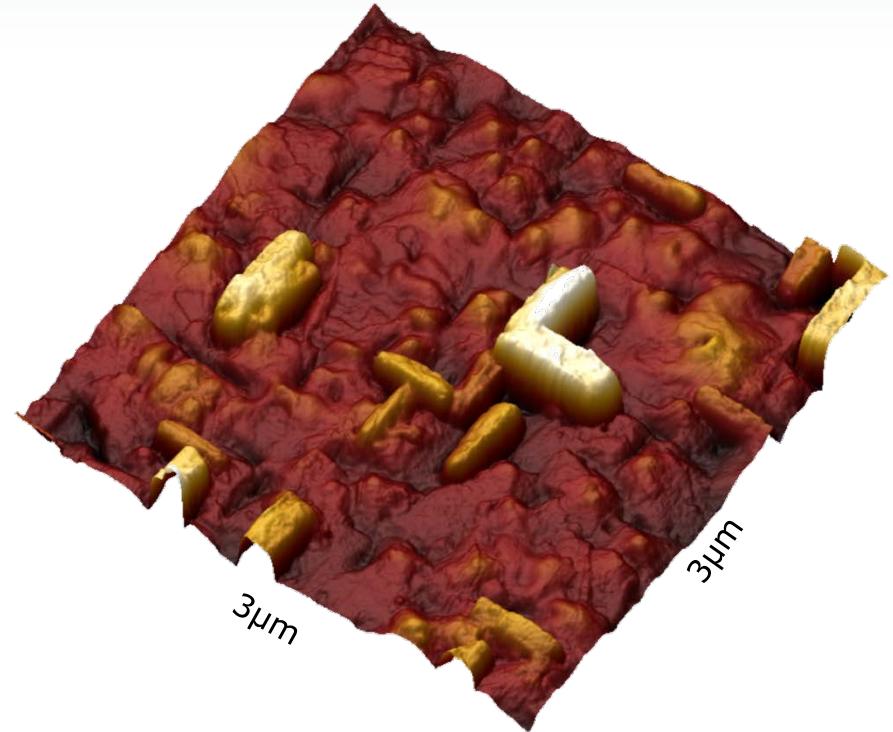


$a = 3.8125(1) \text{ \AA}$   
 $b = 3.8750(2) \text{ \AA}$   
 $c = 11.6250(5) \text{ \AA}$

SrTiO<sub>3</sub>



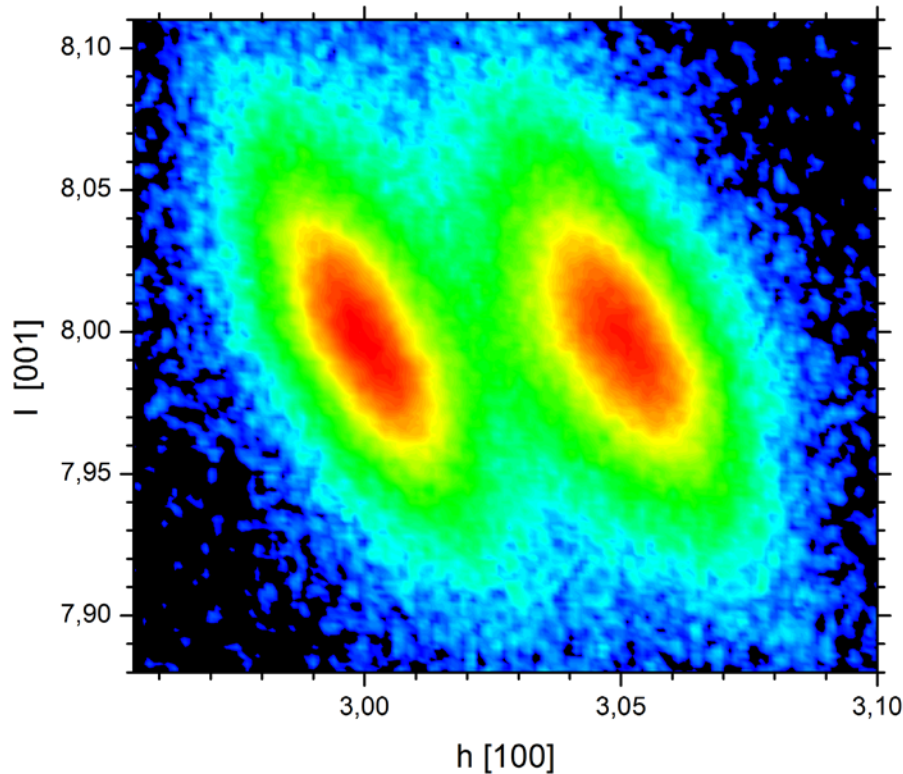
$a = 3.91 \text{ \AA}$



# Co-planar reciprocal space map around YBCO(308)



## YBCO(308+)



- RSM around a **co-planar reflection** YBCO(308+) shows 2 different in-plane lattice parameters.
- Relative lattice mismatch:
$$\frac{\Delta h}{h} = \frac{0.05}{3} \approx 1.7\%$$
- No information about **domain twist**.

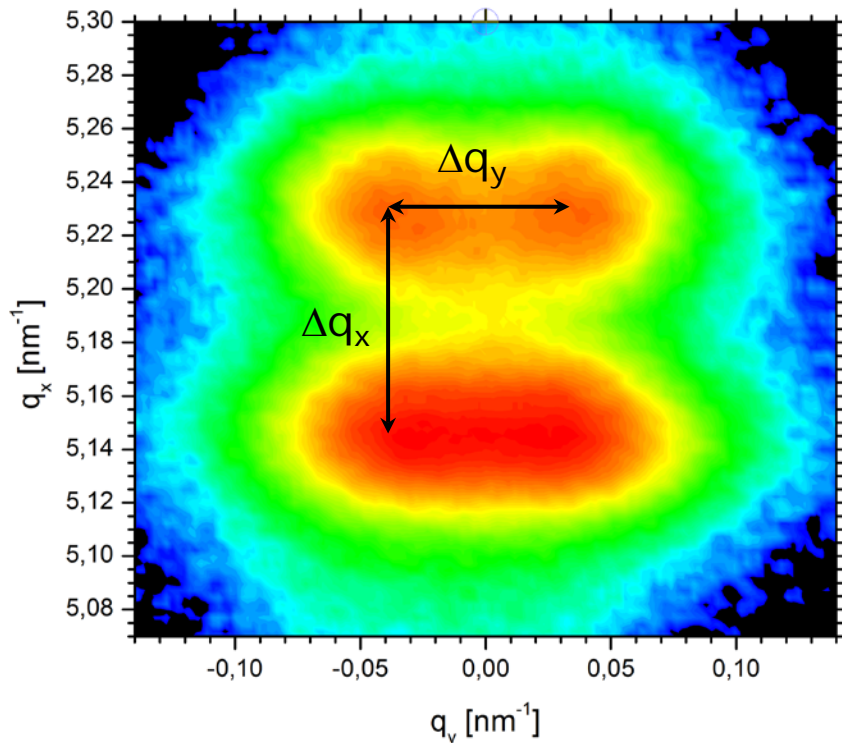


# YBCO on STO

## Probing in-plane symmetry directly



### In-plane RSM @ YBCO(200)



- RSM shows orthorhombic structure with 2 domain orientation.

$$\Delta q_x \approx 0.083 \text{ nm}^{-1}$$

$$\Delta q_y \approx 0.072 \text{ nm}^{-1}$$

- Relative lattice mismatch:

$$\frac{\Delta q_x}{q_x} = 1.63\%$$

- Twist angle of domains:

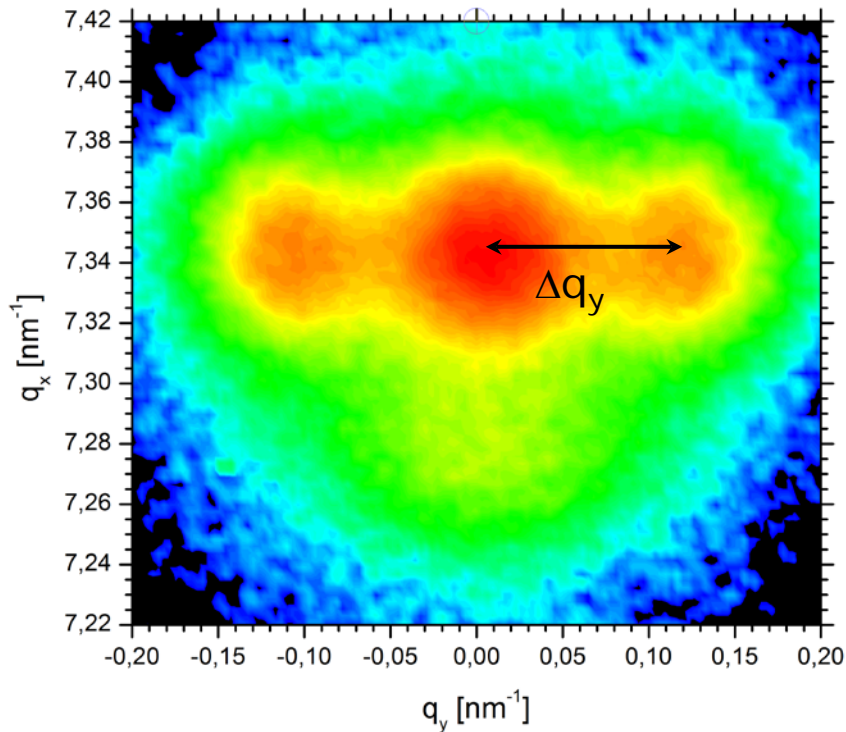
$$\frac{\Delta q_y}{q_x} = 0.8^\circ$$

# YBCO on STO

## Probing in-plane symmetry directly



### In-plane RSM @ YBCO(220)



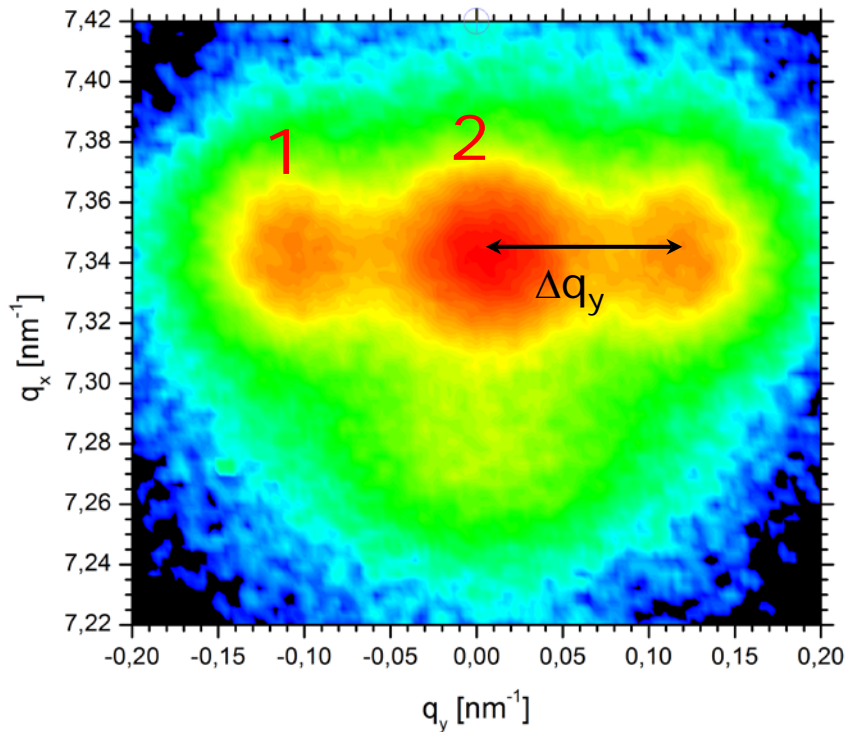
- Twist angle of domains:  $\frac{\Delta q_y}{q_x} = 0.85^\circ$

# YBCO on STO

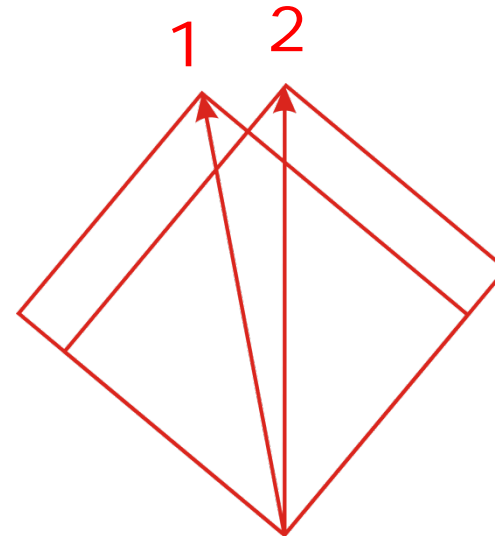
## Probing in-plane symmetry directly



### In-plane RSM @ YBCO(220)



- Twist angle of domains:  $\frac{\Delta q_y}{q_x} = 0.85^\circ$
- Explanation of the RSM

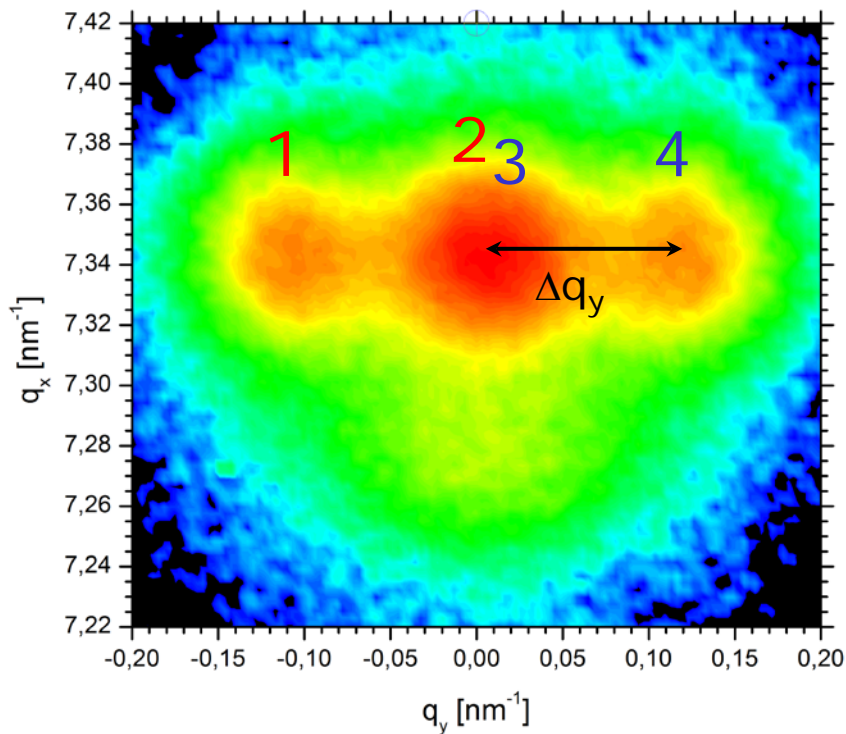


# YBCO on STO

## Probing in-plane symmetry directly

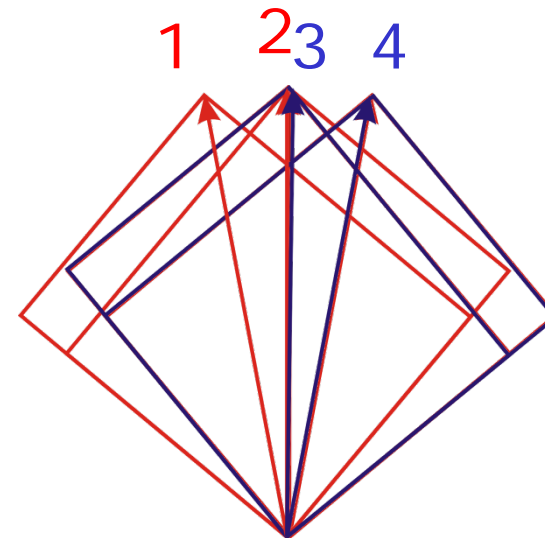


### In-plane RSM @ YBCO(220)



- Twist angle of domains:  $\frac{\Delta q_y}{q_x} = 0.85^\circ$

- Explanation of the RSM

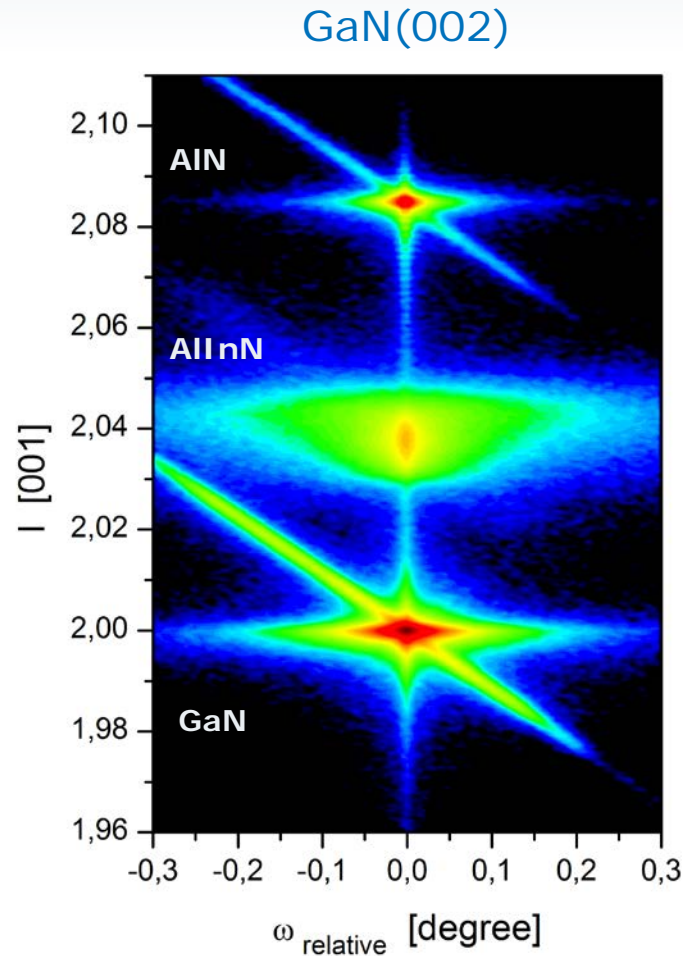


# Example: GaN-based HEMT structure



200nm $\text{Al}_{0.85}\text{In}_{0.15}\text{N}$
1nm $\text{AlN}$
1000nm $\text{GaN}$
350nm $\text{AlN}$
substrate $\text{Al}_2\text{O}_3$

Sample courtesy of L. R. Khoshroo (RWTH Aachen)

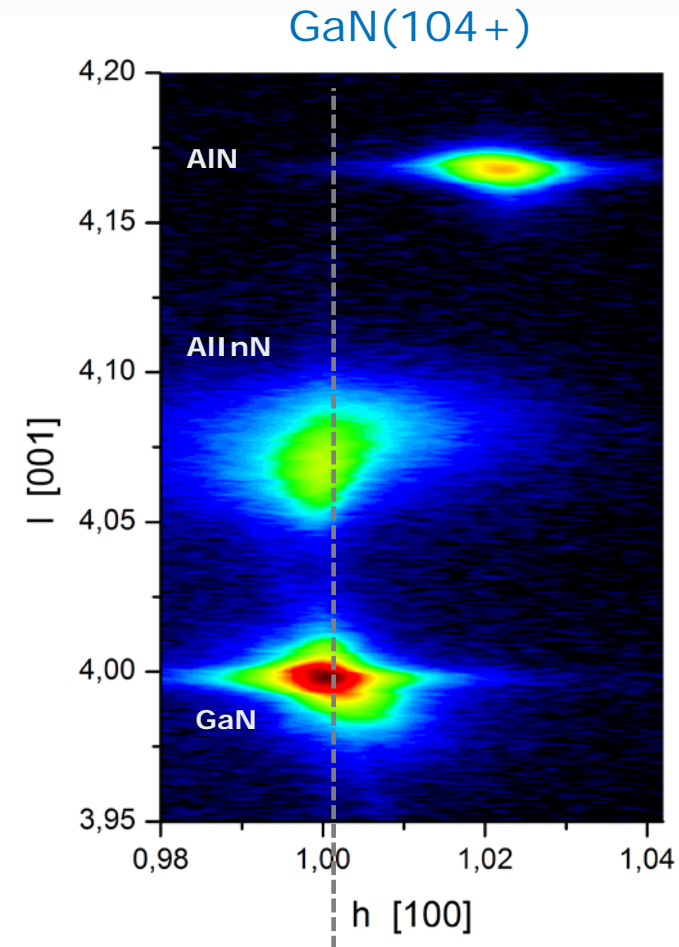
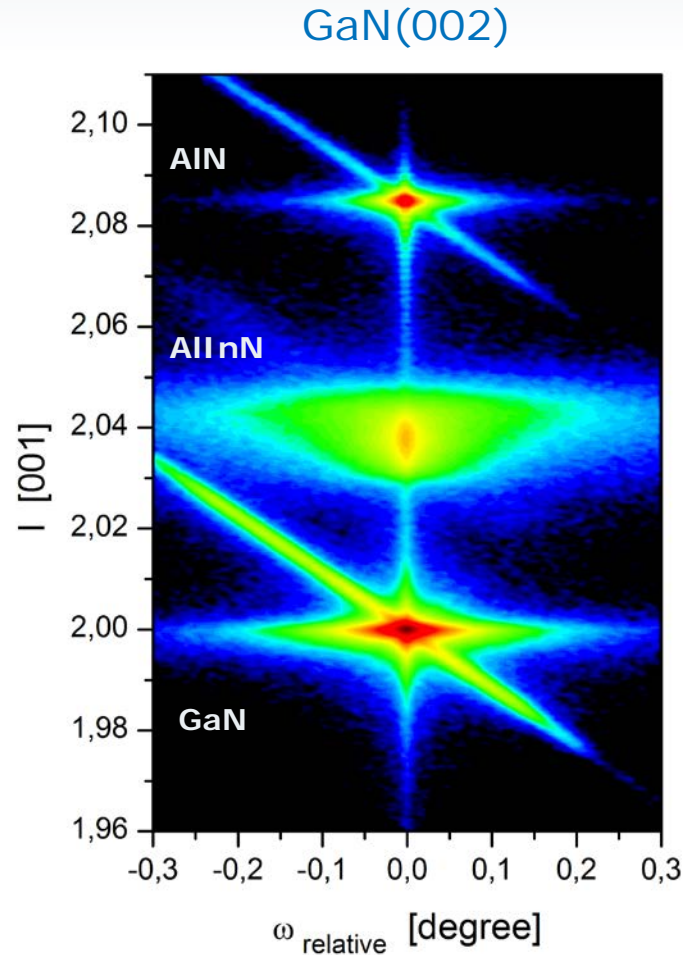


# Example: GaN-based HEMT structure



200nm $\text{Al}_{0.85}\text{In}_{0.15}\text{N}$
1nm $\text{AlN}$
1000nm $\text{GaN}$
350nm $\text{AlN}$
substrate $\text{Al}_2\text{O}_3$

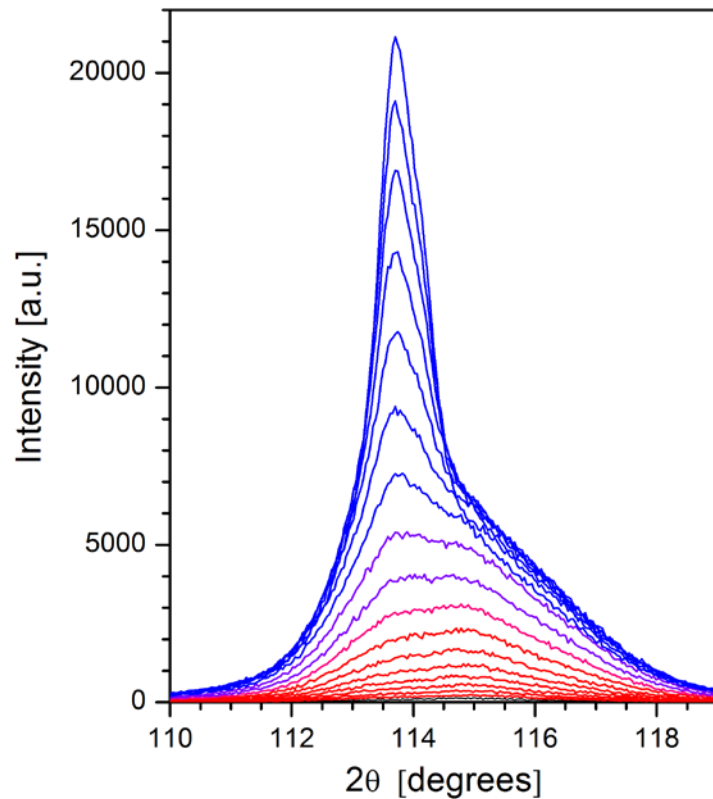
Sample courtesy of L. R. Khoshroo (RWTH Aachen)



# Depth-dependent in-plane GID



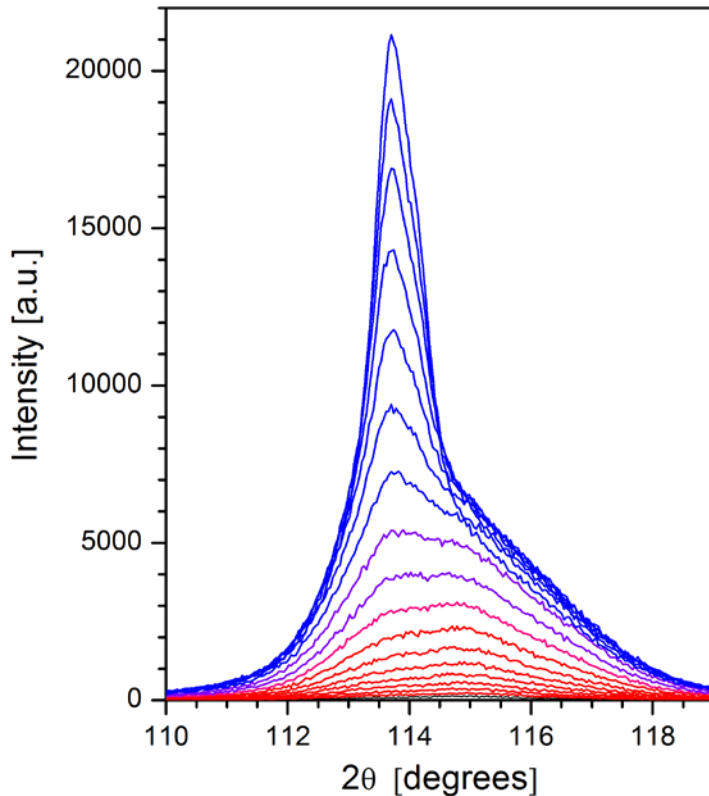
- $2\theta/\omega$  scans at different  $\alpha_i$  around  $\text{Al}_x\text{In}_{1-x}\text{N}(300)$  reflection



# Depth-dependent in-plane GID



- $2\theta/\omega$  scans at different  $\alpha_i$  around  $\text{Al}_x\text{In}_{1-x}\text{N}(300)$  reflection

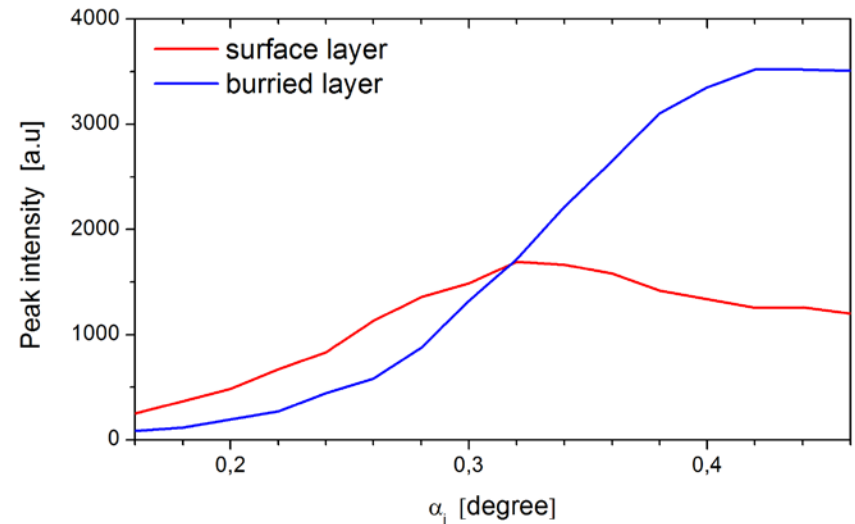


- Peak position obtained with

$$115.04^\circ \pm 0.2^\circ \quad a = 3,162 \text{ \AA}$$

$$113.65^\circ \pm 0.1^\circ \quad a = 3,188 \text{ \AA}$$

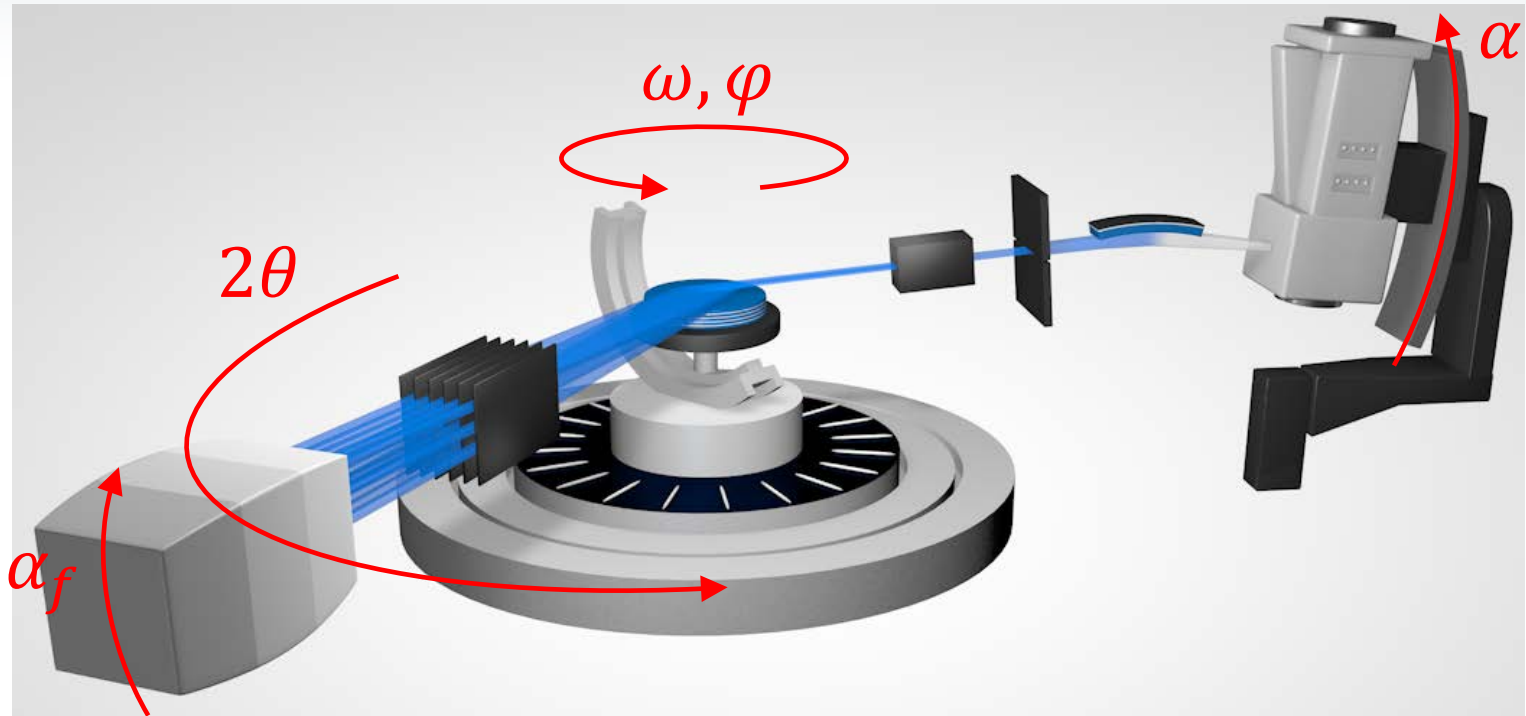
- Integrated peak intensities





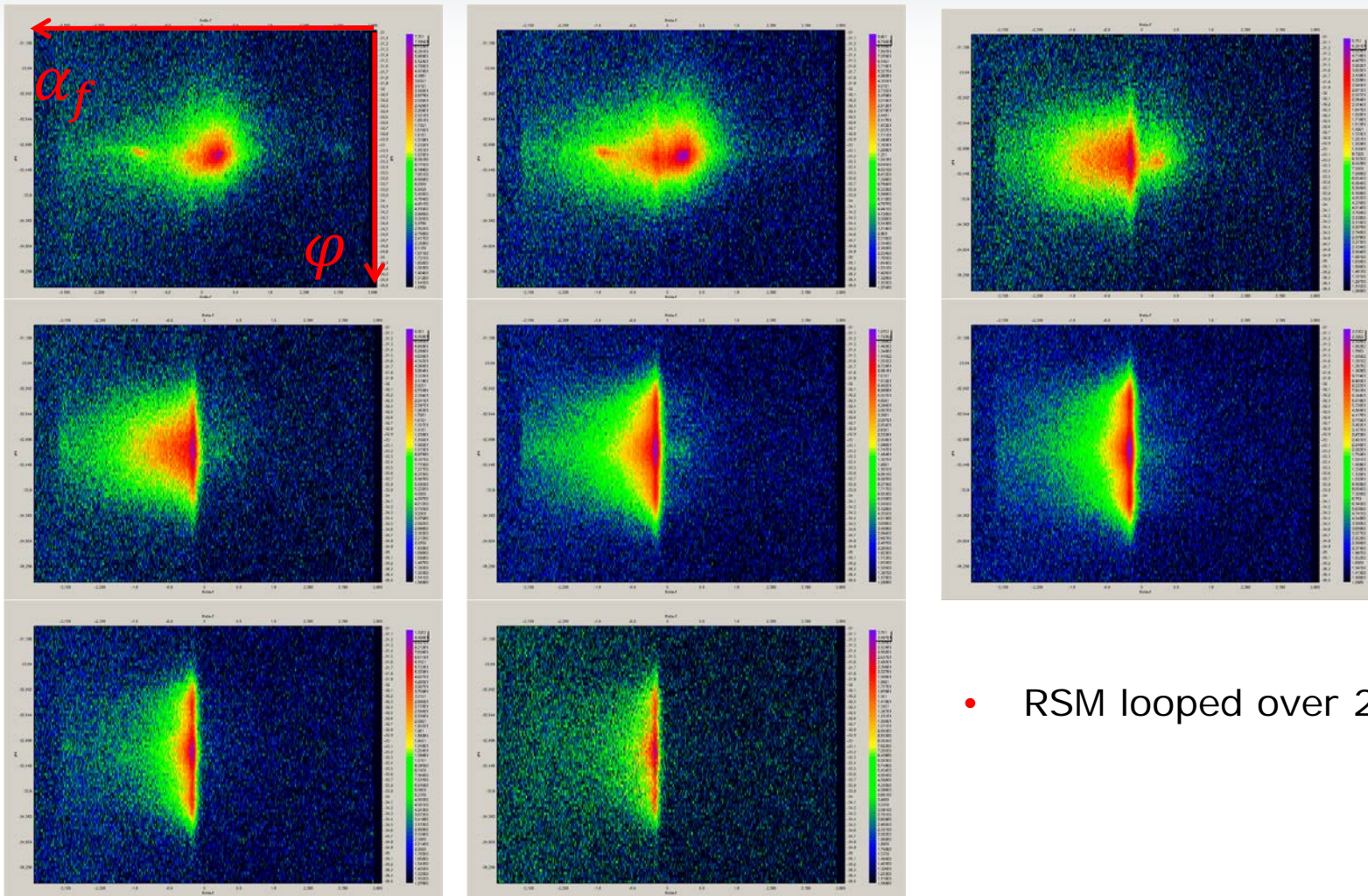
- Introduction
- Experimental configurations
- Performing an experiment
- **Examples**
  - Polycrystalline samples
  - Epitaxial grown samples
  - **In-plane diffraction with 1D detector**

# Ultra-GID configuration



- Use of a 1D-detector rotated by  $90^\circ$  provides resolution perpendicular to the sample surface.

# 3D reciprocal space mapping in IPGID geometry : YBCO(220) on STO



- RSM looped over  $2\theta/\omega$ .

IPGID provides information about

- Crystallite size
- In-plane texture
- In-plane lattice parameter
- Epitaxial relation
- Domain formation and twist
- Depth-dependent information
- Micro strain

Thank you for your attention...



## Any Questions?

Please type any questions you may have in the Q&A panel and then click Send.



[www.bruker.com](http://www.bruker.com)